U	nited S	tates Patent [19]	[11]	Patent	Number:	4,734,328
Ko	hyama et	al.	[45]	Date o	of Patent:	Mar. 29, 1988
[54]	COPOLYN COMPOSI	LINE RANDOM PROPYLENE MER COMPOSITION AND TE LAMINATE COMPRISING MPOSITION	4,528, 4,540, 4,600,	.7/198 .756 9/198 .762 7/198	5 Hwo	
[75]	Inventors:	Masaki Kohyama, Ichihara; Takeshi Muranaka, Iwakuni; Kunisuke Fukui, Hiroshima; Norio Kashiwa, Iwakuni, both of Japan	F 0135 58-032	OREIGN 358 3/198: 611 8/198	PATENT DO 5 European Pa 1 Japan .	OCUMENTS at. Off
[73]	Assignee:	Shell Oil Company, Houston, Tex.	1582 2055		0 United King 1 United King	
[21]	Appl. No.:	29,365			3 United King	
[22]	Filed:	Mar. 23, 1987	_		Carman J. Se irm—Sherma	ccuro n and Shalloway
	Relat	ed U.S. Application Data	[57]		ABSTRACT	
[63]	Continuation doned.	n of Ser. No. 792,670, Oct. 29, 1985, aban-			- -	copolymer composi- andom propylene co-
[30]	Foreign	1 Application Priority Data	polymer c	comprising	99 to 85 mole	% of propylene and
	ov. 2, 1984 [JF ov. 2, 1984 [JF	•	and (ii) a r	andom 1-b	utene copolyn	other than propylene ner comprising ethyl-
[51]	Int. Cl. ⁴	C08L 23/08; C08L 23/20; B32B 27/06; B32B 27/08	to the cop	olymer (ii)	being in the	of the copolymer (i) range of from 5:95 to olymer (ii) satisfying
[52]	U.S. Cl			ing condit	-	01)11101 (11) 04115191115
[58]	Field of Sea	428/516; 525/240 rch 525/240; 428/336, 516	-	_		of the ethylene com- the 1-butene compo-
[56]		References Cited	nent,			tale i outene compo
	U.S. P	ATENT DOCUMENTS			sic viscosity [a	η], measured in deca-
	3,316,226 4/1 3,362,940 1/1 3,489,732 1/1 3,542,717 11/1 3,634,552 1/1 3,699,090 10/1 4,186,240 1/1 4,188,350 2/1 4,230,767 10/1 4,275,119 6/1 4,275,120 6/1	967 Clark et al. 260/88.2 968 Edwards et al. 260/88.2 970 Eichenbaum 260/80.78 970 Lipman 260/27 972 Foglia et al. 525/240 972 Nagel 260/88.2 980 Matsuda et al. 428/349 980 Vicik et al. 428/516 981 Weiner 428/516 981 Weiner 428/516 981 Weiner 428/516 981 Weiner 428/516	ential (D) it h tomet (E) the a it whi more (F) the a it whi	scanning of as a crystal ry, of from mount [With is solution 12%, amount [With is solution and negation is solution.	calorimeter, of linity, measure 1 to 60%, in % by weighted in boiling and 2 in % by weighted in a 1:1 by	measured by a differ- f from 30° to 130° C., red by X-ray diffrac- ght] of that portion of methyl acetate is not ght] of that portion of y volume mixture of 0° C. is less than
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26 Claims, No Drawings

4,384,089 5/1983 Dehm 526/159

CRYSTALLINE RANDOM PROPYLENE COPOLYMER COMPOSITION AND COMPOSITE LAMINATE COMPRISING SAID COMPOSITION

This application is a continuation, of application Ser. No. 792,670 filed Oct. 29, 1985, now abandoned.

BACKGROUND OF THE INVENTION

1. Field of the Invention

This invention relates to a crystalline random propylene copolymer composition which can give a polypropylene composite laminate having improved heat sealing property by laminating it to the surface of a crystalline polypropylene substrate, and to a laminated composite so obtained.

2. Description of the Prior Art

Crystalline polypropylene films have found extensive use in the field of packaging, especially food packaging, because of their superiority in mechanical properties such as tensile strength, rigidity, surface hardness and impact strength, optical properties such as gloss and transparency, and food hygiene such as the freedom from toxicity and odor. They, however, have the defect that temperatures at which a single layer of polypropylene film can be heat-sealed are high and a proper range of these temperatures is narrow.

In an attempt to improve the heat sealing property of polypropylene films, some methods have already been proposed in which a low-melting resin is laminated to one or both surfaces of a crystalline polypropylene film.

For example, Japanese Laid-Open Patent Publication No. 65552/1980 discloses a process in which a propylene random copolymer composition composed of a random ethylene/propylene copolymer comprising propylene as a main component and a random propylene/alpha-olefin copolymer comprising propylene as a main component is laminated to a crystalline polypropylene film.

Japanese Laid-Open Patent Publication No. 91665/1980 discloses a process in which a random propylene copolymer composed of an ethylene/propylene random copolymer comprising propylene as a main component and a random 1-butene/ethylene copolymer 45 comprising 1-butene as a main component is laminated to a crystalline polypropylene film.

Japanese Laid-Open Patent Publication No. 106585/1979 discloses a process in which a random propylene copolymer composition composed of a ran-50 dom ethylene/propylene copolymer comprising propylene as a main component, a copolymer of 1-butene and an ethylenically unsaturated monomer comprising 1-butene as a main component and a low-molecular-weight thermoplastic resin is laminated to a crystalline 55 polypropylene film.

U.S. Pat. No. 4,230,767 claiming Convention priority from the above-cited Japanese Laid-Open Patent Publications Nos. 91665/1980 (Japanese Patent Application No. 165137/1978) and 106585/1979 (Japanese Patent 60 Application No. 13932/1978) discloses a random propylene copolymer composition which can embrace a composition composed of a random ethylene/propylene copolymer comprising propylene as a main component and a 1/butene-propylene copolymer containing 1-65 butene as a main component and includes an embodiment in which such a propylene random copolymer is laminated to a crystalline polypropylene film.

SUMMARY OF THE INVENTION

The present inventors have found that a polypropylene composite laminate obtained by laminating a random crystalline copolymer composition composed of a crystalline random propyplene copolymer comprising propylene as a main component and a 1-butene type random copolymer comprising 1-butene as a main component, a minor proportion of a low-molecular-weight polymer component and having specific characteristics on at least one surface of a crystalline propylene substrate layer can be heat-sealed at relatively low temperatures of a broad range and has excellent heat seal strength, and that the degree of increase of temperatures at which the polypropylene composite laminate can be heat-sealed by low-temperature heat-treatment for obtaining antistatic property is small.

It is an object of this invention therefore to provide a crystalline random copolymer composition, which when laminated to at least one surface of a crystalline polypropylene substrate layer, can give a polypropylene composite laminate having excellent low-temperature heat-sealability and heat seal strength, and a composite laminates so obtained.

According to the invention, there is provided a crystalline random propylene copolymer composition comprising (i) a crystalline random propylene copolymer comprising of 99 to 85 mole % of propylene and 1 to 15 mole % of an alpha-olefin other than propylene and (ii) a random 1-butene copolymer comprising ethylene and 1-butene, the weight ratio of the copolymer (i) to the copolymer (ii) being in the range of from 5:95 to 90:10, said random 1-butene copolymer (ii) satisfying the following conditions:

- (A) it comprises 1 to 50 mole % of the ethylene component and 50 to 99 mole % the 1-butene component,
- (B) it has intrinsic viscosity [η], measured in decalin at 135° C., of from 0.5 to 6 dl/g,
- (C) it has a melting point [Tm], measured by a differential scanning calorimeter, of from 30° to 130° C.
- (D) it has a crystallinity, measured by X-ray diffractometry, of from 1 to 60%,
- (E) the amount [W₁ in % by weight] of that portion of it which is soluble in boiling methyl acetate is not more than 2%, and
- (F) the amount [W₂ in % by weight] of that portion of it which is soluble in a 1:1 by volume mixture of acetone and n-decane at 10° C. is less than $5 \times [\eta]^{-1.2}$.

According to this invention, there is also provided a polypropylene composite laminate obtained by laminating a layer of the above crystalline random propylene copolymer to at least one surface of a substrate layer of crystalline polypropylene.

DETAILED DESCRIPTION OF THE INVENTION

The crystalline random propylene copolymer (i) constituting the crystalline random propylene copolymer composition of this invention is a crystalline random copolymer composed of propylene as a main component and an alpha-olefin other than propylene. The content of propylene in the crystalline random propylene copolymer is usually 99 to 85 mole %, preferably 98 to 90 mole %. The contents of propylene and the other alpha-olefins can, for example, be measured by ¹³C-NMR. The content of the alpha-olefin other than

propylene is usually 1 to 15 mole %, preferably 2 to 10 mole %. The alpha-olefin other than propylene is preferably ethylene and alpha-olefins having 4 to 20 carbon atoms, especially 1-butene, 1-pentene, 1-hexene and 4-methyl-1-pentene. It may be a mixture of at least two 5 of these alpha-olefins.

The crystalline random propylene copolymer (i) has an intrinsic viscosity [η], measured in decalin at 135° C., of usually 1.5 to 4, preferably 1.7 to 3.5, dl/g, a melting point [Tm], measured by a differential scanning calo- 10 rimeter, of usually 120° to 155° C., preferably 120° to 145° C., and a crystallinity, measured by X-ray diffractometry, of preferably 35 to 50%.

The 1-butene random copolymer (ii) as the other component of the crystalline random copolymer com- 15 position of this invention is a 1-butene random copolymer containing 1-butene as a main component.

The random 1-butene copolymer has a 1-butene content of 50 to 99 mole %, preferably 60 to 99 mole %, and an ethylene content of 1 to 50 mole %, preferably 1 to 20 40 mole % [condition (A)]. The content of 1-butene or ethylene can be measured by ¹³C-NMR. As required, the 1-butene random copolymer may contain a small amount of another alpha-olefin such as propylene. If the 1-butene random copolymer contains less than 50 mole 25 % of the 1-butene, a polypropylene composite laminate obtained by laminating the crystalline random propylene copolymer undergoes blocking or has reduced scratch resistance. Furthermore, temperatures at which it can be heat-sealed rise greatly. Furthermore, if the 30 1-butene component content is higher than 99 mole %, the heat-sealing temperature for the polypropylene composite laminate becomes high, and its heat seal strength is reduced.

The intrinsic viscosity $[\eta]$ of the random 1-butene 35 copolymer measured in decalin at 135° C. is 0.5 to 6 gl/g, preferably 1 to 5 dl/g. If the intrinsic viscosity of the 1-butene copolymer is larger than 6 dl/g, it is difficult to reduce the thickness of the heat sealing layer of the polypropylene composite laminate obtained by laminating the crystalline random propylene copolymer composition. If it is less than 0.5 dl/g, the heat seal strength of the composite laminate is reduced, and temperatures at which it can be heat-sealed by heat-treatment rise greatly.

The melting point [Tm] of the 1-butene random copolymer measured by a differential scanning calorimeter is 30° to 130° C., preferably 40° to 120° C. If the melting point of the 1-butene copolymer is higher than 130° C., the heat sealing temperature of a polypropylene 50 composite laminate obtained by laminating the crystalline propylene random copolymer composition becomes high, and its heat seal strength is reduced. If it is lower than 30° C., the polypropylene composite laminate undergoes blocking and has reduced scratch resistance. Furthermore, temperatures at which it can be heat-sealed by heat-treatment rise greatly.

The melting point by a differential scanning calorimeter (DSC melting point) is measured as follows: A 0.1 mm-thick press sheet taken 20 hours after its molding is 60 subjected to differential scanning calorimetry at a temperature raising rate of 10° C./min. over a range of 0° to 200° C., and the maximum endothermic peak is determined and defined as Tm.

The crystallinity (D) of the 1-butene random copoly-65 mer measured by X-ray diffractometry is in the range of 1 to 60%, preferably 1 to 55%. If the crystallinity of the 1-butene random copolymer is higher than 60%, the

heat sealing temperature for a polypropylene composite laminate obtained by laminating the crystalline random propylene copolymer composition becomes high, and its heat seal strength is reduced. If it is less than 1%, the polypropylene composite laminate undergoes blocking and has reduced scratch resistance. Furthermore, temperatures at which it can be heat-sealed by heat-treatment rise greatly. The crystallinity is determined by X-ray diffractometry using a 1.5 mm thick press sheet taken 20 hours after its molding.

The amount $[W_1 \text{ in } \% \text{ by weight}]$ (E) of that portion of the 1-butene random copolymer soluble in boiling methyl acetate is not more than 2% by weight, for example 0.01 to 2% by weight, preferably 0.02 to 1% by weight, especially preferably 0.03 to 0.5% by weight. The amount $[W_1]$ is preferably within the range represented by the following formulae.

 $0.01 \le W_1 \le 0.03a + 0.5$

more prefrably

 $0.02 \le W_1 \le 0.02a + 0.45$

especially preferably

 $0.03 \le W_1 \le 0.015a + 0.4$

In the above formulae, a represents the content in mole % of the ethylene component in the 1-butene copolymer.

If the [W₁] is larger than 2% by weight, a polypropylene composite laminate obtained by laminating the crystalline propylene random copolymer undergoes blocking and has reduced scratch resistance. Furthermore, temperatures at which it can be heat-sealed by heat-treatment rise greatly. The [W₁] is measured by the following method.

A polymer sample, having a size of about 1 mm×1 mm×1mm is put in a cylindrical glass filter, and extracted for 7 hours by a Soxhlet extractor at a reflux frequency of about 1 per 5 minutes. The extraction residue is dried in a vacuum dryer (degree of vacuum less than 10 mmHg) and its weight is measured. The weight of that portion which dissolves in boiling methyl acetate is determined from a weight difference from the original sample. [W₁] is the percentage of the weight of the boiling methyl acetate-soluble portion based on the weight of the original sample.

The amount $[W_2 \text{ in } \% \text{ by weight}]$ (F) of that portion of the 1-butene type copolymer which dissolves at 10° C. in a mixture of acetone and n-decane is less than $5 \times [\eta]^{-1.2}\%$ by weight, for example, $0.1 \times [\eta]^{-1.2}$ to $5 \times [\eta]^{-1.2}\%$ by weight, preferably $-0.2 \times [\eta]^{-1.2}$ to $4.5 \times [\eta]^{-1.2}\%$ by weight, especially preferably $0.33 \times [\eta]^{-1.2}\%$ by weight, based on the weight of the copolymer. $[\eta]$ used herein means the intrinsic viscosity value of the copolymer. If $[W_2]$ is larger than $5 \times [\eta]^{-1.2}$, a polypropylene composite laminate obtained by laminating the crystalline random propylene copolymer composition undergoes blocking and has reduced scratch resistance. Furthermore, temperatures at which it can be heat-sealed by heat-treatment increase. The $[W_2]$ value is determined as follows:

One gram of a copolymer sample, 0.05 g of 2,6-ditert-butyl-4-methylphenol and 50 ml of n-decane are put in a 150 ml flask equipped with stirring vanes and dissolved on an oil bath at 120° C. After the dissolving, the

solution is allowed to cool spontaneously at room temperature for 30 minutes. Then, 50 ml of acetone is added over 30 seconds, and the solution is cooled on a water bath at 10° C. for 60 minutes. The precipitated copolymer is separated from the solution containing a low-5 molecular-weight polymer component by filtration through a glass filter. The solution is dried at 150° C. and 10 mmHg until its weight becomes constant. The weight of the dried product is measured. The amount of the copolymer which dissolves in the mixed solvent is 10 calculated as the percentage of the weight of the original sample copolymer. In the above method, the stirring is continuously effected from the time of dissolution until immediately before the filtration.

The 1-butene random copolymer satisfies a combination of parameters represented by the characteristic
values (A) to (F), and preferably also satisfies the following parameters (G) to (L).

The stress at yield point (G) of the 1-butene random copolymer, measured by the method of JIS K-7113, is 1 20 to 200 kg/cm², preferably 2 to 180 kg/cm². The stress at break point (H), of the 1-butene random copolymer, measured by the method of JIS K-7113, is 3 to 1000 kg/cm², preferably 5 to 800 kg/cm². The elongation at break point of the 1-butene random copolymer, mea- 25 sured by the method of JIS K-7113, is at least 300%, preferably 350 to 1000%. The stress at yield point (G), the stress at break point (H) and the elongation at break point (I) are measured by the testing method of JIS K-7113. Specifically, a 1 mm-thick press sheet is molded 30 by the method of JIS K-6758, and 19 hours after the molding, a test sample (No. 2 in accordance with JIS K-7113) is punched out from the press sheet. The above properties of the test sample were measured at a stretching speed of 50 mm/min. in an atmosphere kept at 25° C. 35 for 20 hours after the molding of the press sheet. When no clear yield point appears, a stress at 20% stretch is taken as the stress at yield point.

The torsional rigidity (J) of the 1-butene random copolymer, measured by the mehod of JIS K-6745, is, 40 for example, 5 to 3000 kg/cm², preferably 10 to 2000 kg/cm². The torsional rigidity is measured as follows: A 1 mm-thick press sheet is molded in accordance with JIS K6758, and 9 days after the molding, a rectangular test sample, 64 mm long and 635 mm wide, is punched 45 out from the press sheet. Ten days after the molding of the press sheet, a load is applied to the test sample in an atmosphere kept at 25° C. at a torsional angle of 50 to 60 degrees, and 5 seconds later, the rigidity of the sample is measured.

The Young's modulus (K) of the 1-butene random copolymer, measured in accordance with the method of JIS K-7113, is 10 to 5000 kg/cm², preferably 20 to 4000 kg/cm². Preferably, the Young's modulus (K) of the 1-butene random copolymer is expressed by the 55 following formula in relation to the ethylene content (b mole %) of the copolymer.

$$5000 \times 10^{-\frac{b}{25}} \ge K \ge 2000 \times 10^{-\frac{b}{15}}$$

The Young's modulus is measured by the same tensile test as in testing (G), (H) and (I).

The standard deviation σ (L) of the 1-butene content of the 1-butene random copolymer is usually not more 65 than 0.6a mole %, preferably not more than 0.4a mole % (a represents the ethylene content in mole % of the 1-butene random copolymer). The standard deviation σ

is a measure of the randomness of the 1-butene random copolymer. A copolymer of this invention which satisfies the characteristic value (L) as well as the characteristic values (A) to (K) shows better properties.

The standard deviation value σ of the 1-butene random copolymer of this invention is determined by the following equation based on the composition distribution of the copolymer. The composition distribution of the copolymer is measured by an extraction-type column fractionation method in which p-xylene is used as a solvent and the extraction temperature is changed stepwise from 0° to 130° C. at 5° C. intervals. Extraction at a given temperature is carried out for 4 hours by using 2 liters of p-xylene per 10 g of the copolymer sample.

$$\sigma = \left[\int_0^1 100 \left(\bar{x} - x \right)^2 f(x) dx \right]^{\frac{1}{2}}$$

wherein \bar{x} represents the average content in mole % of 1-butene in the copolymer, x represents the 1-butene content (mole %), and f(x) represents the differential weight fraction of a component having the 1-butene content x (mole %).

The 1-butene random copolymer (ii) can be prepared, for example, by copolymerizing 1-butene and ethylene at a temperature of about 20° to about 200° C. in the presence of a catalyst and a molecular weight controlling agent such as hydrogen, the catalyst being formed from

- (a) a highly active titanium catalyst component containing magnesium, titanium, halogen and a diester component as essential ingredients and prepared by reacting a magnesium compound, a titanium compound and a diester and optionally a halogen compound (not always necessary when the magnesium or titanium compound contains a halogen atom),
- (b) an organoaluminum compound, and
- (c) an organic silicon compound catalyst component having an Si—O—C bond.

In the production of the 1-butene random copolymer, the catalyst and the polymerization conditions may be selected by preliminary experiments so that it satisfies the characteristics (A) to (F).

The titanium catalyst component (a) is prepared preferably by contacting a magnesium compound (or magnesium metal), a titanium compound and a diester or a diester-forming compound with or without other reagents. The preparation can be effected in the same way as in the preparation of known highly active titanium catalyst components containing magnesium, titanium, halogen and an electron donor as essential ingredients. For example, it can be produced in accordance with the methods disclosed in British patent specifications Nos. 1492618, 1554340 and 1554248, U.S. Pat. Nos. 4,157,435, 4,076,924, 4,085,276, 4,250,285, 4,232,139, 4,143,223, 4,315,874, 4,330,649, 4,401,589 and 4,335,015, and European patent specification No. 22675.

Several embodiments of producing the titanium catalyst component (a) will be illustrated below.

(1) A magnesium compound or a complex of a magnesium compound and an electron donor is pre-treated or not-pretreated with an electron donor and/or a reaction aid such as an organoaluminum compound or a halogen-containing silicon compound in the presence or absence of an electron donor or a pulverization aid with

or without pulverization. The resulting solid is reacted with a titanium compound which is in the liquid state under the reaction conditions. In the above procedure, the electron donor is used at least once as the electron donor.

- (2) A magnesium compound in the liquid state having no reducing ability is reacted with a titanium compound in the liquid state in the presence of an electron donor to precipitate a solid titanium complex.
- (3) The product obtained in (2) above is further re- 10 acted with a titanium compound.
- (4) The product obtained in (1) or (2) is further reacted with a titanium compound and an electron donor.
- (5) A magnesium compound or a complex of a magnesium compound and an electron donor is pulverized 15 in the presence of a titanium compound and in the presence or absence of an electron donor and a pulverization aid, and with or without pre-treatment with an electron donor and/or a reaction aid such as an organo-aluminum compound or a halogen-containing silicon 20 compound, treated with a halogen, a halogen compound or an aromatic hydrocarbon. In the above procedure, the electron donor is used at least once.

Preferred among these methods are those in which a liquid titanium halide is used, or a halogenated hydro- 25 carbon is used after or during the use of the titanium compound.

The electron donor used in the above methods of preparation is not necessarily limited to the diester or diester-forming compound. There may be used other 30 electron donors such as alcohols, phenols, aldehydes, ketones, ethers, carboxylic acids, carboxylic acid anhydrides, carbonic acid esters, monoesters and amines.

Preferred examples of the diester as an essential component of the highly active solid titanium catalyst com- 35 ponent (a) used in the invention include diesters of dicarboxylic acids in which two carboxyl groups are attached to one carbon atom and dicarboxylic acids in which one carboxyl group is attached to each of the two adjoining carbon atoms. Examples of the dicarbox- 40 ylic acids in such dicarboxylic acid esters are malonic acid, substituted malonic acid, succinic acid, substituted succinic acid, maleic acid, substituted maleic acid, fumaric acid, substituted fumaric acid, alicyclic dicarboxylic acids in which two carboxyl groups are attached to 45 one carbon atom forming the aliphatic ring, alicyclic dicarboxylic acids in which one carboxyl group is bonded to each of the two adjoining carbon atoms forming the aliphatic ring, aromatic dicarboxylic acids having carboxyl groups at the ortho-position, and heter- 50 ocyclic dicarboxylic acids having one carboxyl group to each of the two adjoining carbon atoms forming the heterocyclic ring.

Specific examples of the dicarboxylic acids exemplified above include malonic acid; substituted malonic 55 acids such as methylmalonic acid, ethylmalonic acid, isopropylmalonic acid, allylmalonic acid, and phenylmalonic acid; succinic acid; substituted succinic acids such as methylsuccinic acid, dimethylsuccinic acid, ethylsuccinic acid, methylethylsuccinic acid and ita-60 conic acid; maleic acid; substituted maleic acids such as citraconic acid and dimethylmaleic acid; alicyclic dicarboxylic acid, cyclopentane-1,2-dicarboxylic acid, cyclohexane-1,2-dicarboxylic acid, cyclohexene-1,2-dicarboxylic acid, cyclohexene-3,4-dicarboxylic acid, cyclo-hexene-4,5dicarboxylic acid, Nadic Acid, Methylnadic Acid, and 1-allylcy-

clohexane-3,4-dicarboxylic acid; aromatic dicarboxylic acids such as phthalic acid, naphthalene-1,2-dicarboxylic acid and naphthalene-2,3-dicarboxylic acid; and heterocyclic dicarboxylic acids such as furane-3,4-dicarboxylic acid, benzopyran-3,4-dicarboxylic acid, pyrrole-2,3-dicarboxylic acid, thiophene-3,4-dicarboxylic acid, and indole-2,3-dicarboxylic acid.

Preferably, at least one of the alcohol components of the dicarboxylic acid diesters exemplified above has at least 2 carbon atoms, especially at least 3 carbon atoms. It is above all preferred that both of the alcohol components have at least 2 carbon atoms, especially at least 3 carbon atoms. Examples include the diethyl esters, diisopropyl esters, di-n-propyl esters, di-n-butyl esters, diisobutyl esters, di-tert-butyl esters, diisoamyl esters, di-n-hexyl esters, di-2-ethylhexyl esters, di-n-octyl esters, diisodecyl esters, and ethyl n-butyl esters of the above-exemplified dicarboxylic acids.

Both a magnesium compound having reducing ability and a magnesium compound having no reducing ability can be utilized in the preparation of the solid highly active titanium catalyst component (a).

The former includes, for example, magnesium compounds having a magnesium-carbon bond or a magnesium-hydrogen bond, for example dimethyl magnesium, diethyl magnesium, dipropyl magnesium, dibutyl magnesium, ethylbutyl magnesium, diamyl magnesium, dihexyl magnesium, didecyl magnesium, ethylmagnesium chloride, propylmagnesium chloride, butylmagnesium chloride, hexylmagnesium chloride, amylmagnesium chloride; ethyl butylmagnesium and butylmagnesium hydride. These magnesium compounds may be used in the form of a complex with an organoaluminum compound, for example, or may be in the form of a liquid or a solid.

The latter includes, for example, magnesium halides such as mangnesium chloride, magnesium bromide, magnesium iodide and magnesium fluoride; alkoxymagnesium halides such as methoxymagnesium chloride, ethoxymagnesium chloride, isopropoxymagnesium chloride, butoxymagnesium chloride and octoxymagnesium chloride; aryloxymagnesium halides such as phenoxy magnesium chloride and methylphenoxymagnesium chloride; alkoxymagnesiums such as ethoxy magnesium, isopropoxy magnesium, butoxy magnesium, n-octoxy magnesium, and 2-ethylhexoxy magnesium; aryloxymagnesiums such as phenoxy magnesium and dimethylphenoxy magnesium; and carboxylic acid salts of magnesium such as magnesium laurate and magnesium stearate. These magnesium compounds having no reducing ability may be derived from the aforesaid magnesium compounds having reducing ability, or those derived during the preparation of the catalyst component. The above magnesium compounds may be a complex with other metals or mixtures of other metal compounds. Or they may be a mixture of two or more of these compounds.

Preferred are the magnesium compounds having no reducing ability, and halogen-containing magnesium compounds, particularly, magnesium chloride, alkoxy magnesium chlorides and aryloxymagnesium chlorides are preferred.

Suitable titanium compounds used to prepare the titanium catalyst component (a) are tetravalent titanium compounds represented by $Ti(OR)_g x_{4-g}$ in which R is a hydrocarbon group, X is halogen and g is 0 to 4.

Specific examples of such titanium compounds include titanium tetrahalides such as TiCl4, TiBr4 and TiI₄; alkoxytitanium trihalides such as Ti(OCH₃)Cl₃, Ti(OC₂H₅)Cl₃, Ti(O n- C₄H₉)Cl₃, Ti(OC₂H₅)Br₃ and Ti(O iso-C₄H₉)Br₃; alkoxytitanium dihalides such as Ti(OCH₃)₂Cl₂, Ti(OC₂H₅)₂Cl₂, Ti(O N-C₄H₉)₂Cl₂ and Ti(OC₂H₅)₂Br₂; trialkoxytitanium monohalides such as Ti(OCH₃)₃Cl, Ti(OC₂H₅)₃Cl, Ti(O n-C₄H₉)₃Cl and Ti(OC₂H₅)₃Br; and tetraalkoxytitaniums such as Ti- $(OCH_3)_4$, $Ti(OC_2H_5)_3$ and $Ti(O n-C_4H_9)_4$. Among 10 them, the halogen-containing titanium compounds, particularly titanium tetrahalides, especially preferably titanium tetrachloride, are preferred. These titanium compounds may be used singly or as a mixture. Or they may be used as diluted in hydrocarbons or halogenated 15 hydrocarbons.

In the preparation of the titanium catalyst component (a), the amounts of the titanium compound, the magnesium compound the electron donor to be supported, and the other electron donors such as alcohols, phenols, monocarboxylic acid esters, the silicon compound-and the aluminum compound which may used as required differ depending upon the method of preparation and cannot be defined in a general manner. For example, about 0.1 to about 10 moles of the electron donor to be supported and about 0.05 mole to about 1000 moles of the titanium compound may be used per mole of the magnesium compound.

In the present invention, a catalyst composed of the solid highly active titanium catalyst component (a) described above, (b) an organoaluminum compound and (c) an organic silicon compound having an Si—O—C bond is used.

Examples of the organoaluminum compound (a) include (1) organoaluminum compounds at least having an Al-carbon bond in the molecule, for example organoaluminum compounds represented by the general formula

$$R^1_m Al(OR^2) _n H_p X_q$$

wherein each of R^1 and R^2 , which may be identical or different, represents a hydrocarbon, for example a C_1 - C_{15} hydrocarbon group, preferably C_1 - C_4 hydrocarbon group, X represents a halogen atom such as 45 $0 \le m \le 3$, $0 \le n \le 3$, $0 \le p \le 3$, $0 \le q \le 3$, provided that m+n+p+q=3, and

(2) complex alkylated products of aluminum and a metal of Group I of the periodic table represented by the following general formula

 $M^1AIR^1_4$

wherein M¹ is Li, Na or K, and R¹ is as defined above. Examples of the organoaluminum compounds (a) are 55 those of the general formulae

 $R^1_m Al(OR^2)_{3-m}$

wherein R^1 , R^2 and X are as defined above, and m is $_{60}$ preferably a number represented by 0 < m < 3.

 $R^1_mAlH_{3-m}$

wherein R_1 is as defined above, and m is preferably a on number represented by $2 \le m < 3$,

 $R^1_m Al(OR^2)_n X_q$

wherein R^1 , R^2 and X are as defined above, $0 < m \le 3$, $0 \le n < 3$, $0 \le q < 3$, m+n+q=3.

Examples of the organoaluminum compounds (a) include trialkyl aluminums such as triethyl aluminum and tributyl aluminum; trialkenyl aluminums such as triisoprenyl aluminum; dialkyl aluminum alkoxides such as diethyl aluminum ethoxide and dibutyl aluminum butoxide; partially alkoxylated alkyl aluminums such as alkyl aluminum sesquialkoxides (e.g., ethyl aluminum sesquiethoxide and butylaluminum sesquibutoxide) and partially alkoxylated alkyl aluminums having the average composition represented by R¹_{2.5}Al(OR²)_{0.5}; dialkyl aluminum halides such as diethyl aluminum chloride; dibutyl aluminum chloride and diethyl aluminum bromide; alkyl aluminum sesquihalides such as ethyl aluminum sesquichloride, butyl aluminum sesquichloride and ethyl aluminum sesquibromide; partially halogenated alkyl aluminums, for example alkyl aluminum dihalides such as ethyl aluminum dichloride, propyl aluminum dichloride and butyl aluminum dibromide; other partially halogenated alkyl aluminums, for example alkyl aluminum dihydrides such as ethyl aluminum dihydride and propyl aluminum dihydride; and partially alkoxylated and halogenated alky aluminums such as ethyl aluminum ethoxychloride, ethyl aluminum butoxy chloride and ethyl aluminum ethoxy bromide.

LiAl(C₂H₅)₄ and LiAl(C₇H₁₅)₄ may be cited as examples of the compounds (b).

There may also be used organic aluminum compounds in which two or more aluminum atoms are bonded through an oxygen or nitrogen atom which are similar to the compounds (1). Examples are $(C_2H_5)_2AlOAl(C_2H_5)_2$, $(C_4H_9)_2AlOAl(C_4H_9)_2$, and

(C₂H₅)AlNal(C₂H₅). | | C₂H₅

Of these, the trialkyl aluminums and the alkyl aluminums in which two or more aluminum atoms are bonded are preferred.

Illustrative of the organic silicon compound (c) having an Si—O—C are alkoxysilanes and aryloxysilanes. For example, there may be cited organic silicon compounds represented by the following general formula

 $R_n Si(OR^1)_{4-n}$

wherein R represents a hydrocarbon group, such as an alkyl, cycloalkyl, aryl, alkenyl, haloalkyl, or aminoalkyl group, or halogen, R¹represents a hydrocarbon group such as an alkyl, cycloalkyl, aryl, alkenyl or alkoxyalkyl group, and n is a number represented by 0≤n≤3, and n R groups, or (4-n)OR¹ groups may be identical or different.

Other examples of the catalyst component (c) include siloxanes having the group OR¹ and silyl esters of carboxylic acid. Compounds in which two or more silicon atoms are bonded to each other through an oxygen or nitrogen atom may be cited as still another example. There may also be used the product of reaction of a compound having no Si—O—C bond with a compound having an O—C bond obtained either in advance or in situ. There can be cited the combined use of a halogen-containing silane compound containing no Si—O—C bond or silicon hydride with an alkoxy group-containing aluminum compound, an alkoxy group-con-

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taining magnesium compound, a metal alcoholate, an alcohol, a formic acid ester, ethylene oxide, etc. The organic silicon compounds may also include other metals such as aluminum or tin.

Specific examples of preferred organic silicon com- 5 pounds as component (c) include trimethylmethoxysilane, trimethylethoxysilane, dimethyldimethoxysilane, dimethyldiethoxysilane, diphenyldimethoxysilane, methylphenyldimethoxysilane, diphenyldiethoxysilane, ethyltrimethoxysilane, methyltrimethoxysilane, vinyl- 10 trimethoxysilane, phenyltrimethoxysilane, gammachloropropyltrimethoxysilane, methyltriethoxysilane, ethyltriethoxysilane, vinyltriethoxysilane, butyltriethoxysilane, phenyltriethoxysilane, gammaaminopropyltriethoxysilane, chlorotriethoxysilane, ethyl- 15 triisopropoxysilane, vinyltributoxysilane, ethyl silicate, butyl silicate, trimethylphenoxysilane, methyltriallyloxysilane, vinyltris(beta-methoxyethoxy)silane, vinyltriacetoxysilane, dimethyltetraethoxydisiloxane and phenyldiethoxydiethylaminosilane. Of these, methyltri- 20 methoxysilane, phenyltrimethoxysilane, methyltriethoxysilane, ethyltriethoxysilane, vinyltriethoxysilane, phenyltriethoxysilane, vinyltributoxysilane, ethyl silicate, diphenyldimethoxysilane, diphenyldiethoxysilane and methylphenylmethoxysilane (the compounds of 25 formula $R_n(OR^1)_{4-n}$ given above in which n is preferably 0 (or 1)) are especially preferred.

The weight ratio of the crystalline random propylene copolymer (i) to the 1-butene random copolymer (ii) in the crystalline random propylene copolymer composi- 30 tion of this invention is from 5:95 to 90:10, preferably from 10:90 to 85:15.

The crystalline random propylene copolymer composition of this invention is composed of the crystalline random propylene copolymer (i) and the 1-butene ran- 35 dom copolymer (ii), or it may contain another polymer in addition to the two polymers (i) and (ii). The copolymer composition of this invention may further include ordinary additives such as a heat stabilizer, a weather stabilizer, a nucleating agent, a lubricant, a slip agent, an antihaze agent, a pigment and a dye. The suitable proportions of these additives are those which do not impair the low-temperature heat-sealability and heat seal strength of a polypropylene composite laminate prepared by using 45 the composition of this invention.

The copolymer composition of this invention may be prepared by uniformly mixing the aforesaid components by a tumbler, a V-type blender, a Henschel mixer, etc. If desired, the resulting mixture may further be 50 kneaded by an extruder, a Banbury mixer, a kneader, a roll, etc.

A polypropylene composite laminate is prepared by laminating the crystalline random propylene copolymer composition of this invention to one or both surfaces of 55 a substrate composed of crystalline polypropylene. The crystalline polypropylene as the substrate is, for example, crystalline propylene homopolymer, or a crystalline random propylene/alpha-olefin copolymer comprising propylene as a main component such as a 60 propylene/ethylene random copolymer having an ethylene content of 0.1 to 5 mole \%, a propylene/ethylene/1-butene random copolymer having an ethylene content of 0.1 to 4 mole % and a 1-butene content of 0.1 to 5 mole %, or a propylene/1-butene random 65 copolymer having a 1-butene content of 0.1 to 5 mole %. The crystalline polypropylene has an intrinsic viscosity $[\eta]$, measured in decalin at 135° C., of usually 1.5

to 4 dl/g, preferably 1.7 to 3.5 dl/g, and a crystallinity, measured by X-ray diffractometry, of usually 50 to 70%, preferably 55 to 70%. The substrate layer composed of the crystalline polypropylene may be non-stretched or in the monoaxially or biaxially stretched state.

The following methods may be cited by way of exemplification for the production of the polypropylene composite laminate.

- (1) A method which comprises co-extruding a substrate composed of crystalline polypropylene and the crystalline random propylene copolymer composition, and as required, stretching the resulting laminate separately or simultaneously in the machine direction and/or the transverse direction.
- (2) A method which comprises extruding the crystalline random propylene copolymer composition in the molten state onto the surface of a non-stretched, monoaxially stretched or biaxially stretched substrate and when the substrate is non-stretched, optionally stretching the resulting laminate monoaxially or biaxially. When the substrate is monoaxially stretched, the resulting laminate may, if required, be further stretched in the same direction as the stretching of the substrate or in a direction crossing that direction.
- (3) A method which comprises laminating a film of the crystalline random propylene copolymer composition to the surface of a substrate of crystalline polypropylene by using an adhesive. Examples of the adhesive used are emulsions of polyvinylidene chloride, polyacrylic acid esters and chlorinated polypropylene. After applying the adhesive to the substrate, the film of the composition is laminated.

The polypropylene composite laminate so prepared may be in any shape, for example a laminated film, a laminated sheet, a laminated packaging bag, a laminated container or another molded article having heat-sealability.

As is clear from the above exemplified methods of lamination, the substrate layer of crystalline polypropylene constituting the propylene composite laminate may be in the non-stretched state or in the monoaxially or biaxially stretched state. The layer of the crystalline random propylene copolymer composition may likewise be non-stretched or monoaxially or biaxially stretched. The substrate layers in the above states and the copolymer composition layers in the above states may be used in any desired combinations.

The thickness of the substrate layer of crystalline polypropylene is not particularly restricted. But the heat sealed layer of the crystalline random propylene copolymer composition has a thickness of generally 0.1 to 50 microns, preferably 0.5 to 30 microns. When the polypropylene composite laminate is a composite laminated film or a composite laminated sheet, the thickness of the substrate layer of crystalline polypropylene is 5 to 200 microns, preferably 10 to 70 microns, and the heat-sealed layer composed of the crystalline random propylene copolymer composition is usually 0.1 to 50 microns, preferably 0.5 to 30 microns.

The polypropylene composite laminate formed by laminating a layer of the crystalline random propylene copolymer composition of this invention to one or both surfaces of a substrate of crystalline polypropylene has excellent low-temperature heat sealability and heat seal strength with a broadened range of temperatures at which it can be heat sealed. In addition, it has superior scratch resistance and antiblocking property. By taking

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advantage of these properties, the laminate finds ruitable applications in packaging of daily goods and sundries such as food packaging and apparel packaging.

The following examples illustrate the present invention specifically. In these examples, the various items 5 shown below were tested by the following methods.

(1) Heat seal strength

That surface of a composite polypropylene film on which the crystalline random propylene copolymer 10 composition is laminated is contacted with itself, and the contacting portion is heat-sealed by means of a seal bar having a width of 5 mm for 1 second under a pressure of 2 kg/cm² at a given temperature and then allowed to cool spontaneously. A test sample, 15 mm 15 width, is cut out from the heat-sealed product, and the strength at which the heat-sealed portion is peeled apart at a crosshead speed of 200 mm/min. is determined and defined as the heat seal strength.

Separately, the above composite polypropylene film 20 is left to stand for 1 week in an air atmosphere at 50° C., and then the strength of the heat-sealed portion is measured by the same method as above and defined as the heat seal strength after heat-treatment.

(2) Scratch resistance

That surface of a composite polypropylene film on which the crystalline random propylene copolymer is laminated is contacted with itself, and the contacting surface portions are rubbed with each other 15 times 30 using an iron block weighing 5 kg as a load. Then, the degree of haze of the sample is measured by the method of (4) described below, and the difference (Δ degree of haze) of the measured degree of haze from the degree of haze of the sample before rubbing is determined.

(3) Antiblocking property

Determined in accordance with ASTM D1893. A sample, 10 cm wide and 15 cm long, is cut out from a composite polypropylene film. That surface of the sam- 40 ple on which the crystalline random propylene copolymer composition is laminated is contacted with itself. The contacting portion is held by two glass sheets. A load of 10 kg is placed on it, and the sample is left to stand in an air oven at 50° C. The sample is taken out 1 45 day and 7 days later, and its peel strength is measured by a universal tester. The peel strength value per cm is defined as a blocking value.

(4) Degree of haze (ASTM D1003)

A film formed is aged in an air oven at 50° C. The degree of haze of the film is measured before the aging and 1 day and 7 days after the aging.

(5) Slipping property (ASTM D1894)

A film prepared is aged in an air oven at 50° C. The coefficient of static friction and the coefficient of dynamic friction of the film are measured before the aging and 1 day and 7 days after the aging.

The production of the 1-butene random copolymers 60 used in the following Examples and Comparative Examples is shown in Referential Examples 1 to 9, and the properties of the resulting copolymers are summarized in Table 1.

Table 2 summarizes the properties of crystalline poly- 65 propylene used as a substrate layer and the crystalline random propylene copolymer (i) incorporated in the crystalline random copolymer composition. These

propylene polymers were used in the following Examples and Comparative Examples.

REFERENTIAL EXAMPLE 1

Preparation of a titanium catalyst component (a)

Anhydrous magnesium chloride (4.76 g; 50 mmoles), 25 ml of decane and 23.4 ml (150 mmoles) of 2-ethylhexyl alcohol were reacted at 130° C. for 2 hours to form a uniform solution. Phthalic anhydride (1.11 g; 7.5 mmoles) was added to the solution, and the mixture was stirred for 1 hour at 130° C. to dissolve phthalic anhydride uniformly in the solution. The resulting uniform solution was cooled to room temperature, and 200 ml (1.8 moles) of titanium tetrachloride kept at -20° C. was added dropwise over 1 hour. After the addition, the temperature of the mixture was raised to 110° C. over 4 hours. When its temperature reached 100° C., 2.68 ml (12.5 mmoles) of diisobutyl phthalate was added, and the mixture was maintained at this temperature for 2 hours with stirring. After the 2-hour reaction, the solid portion was collected by hot filtration, and suspended in 200 ml of TiCl₄. Again, the suspension was reacted at 110° C. for 2 hours. After the reaction, the solid portion was collected by hot filtration, and washed thoroughly with decane and hexane at 110° C. until no free titanium compound was detected from the washings. The titanium catalyst component (a) so prepared was stored as a hexane slurry. A part of it was dried to examine the composition of the catalyst component. The resulting titanium catalyst component (a) contained 3.1% by weight of titanium, 56.0% by weight of chlorine, 17.0% by weight of magnesium and 20.9% by weight of diisobutyl phthalate.

Polymerization

A 200-liter SUS reaction vessel was continuously charged hourly with 50 kg of 1-butene, 0.18 kg of ethylene, 100 mmoles of triethyl aluminum, 10 mmoles of vinyltriethoxysilane and 0.5 mmole, as titanium atom, of the titanium catalyst component (a). The partial hydrogen pressure in the vapor phase was maintained at 1.5 kg/cm², and the polymerization temperature, at 70° C.

The polymer solution was continuously withdrawn so that the amount of the solution in the reaction vessel became 100 liters. A small amount of methanol was added to stop the polymerization, and the unreacted monomers were removed. A butene-1/ethylene copolymer was obtained in an amount of 9.6 kg per hour. The 50 results are shown in Table 1.

REFERENTIAL EXAMPLES 2-7

Referential Example 1 was repeated except that the amount of ethylene charged and the partial hydrogen 55 pressure were changed as indicated in Table 1.

REFERENTIAL EXAMPLE 8

A 200-liter reaction vessel was continuously charged hourly with 50 kg of 1-butene, 0.3 kg of ethylene, 200 mmoles of diethyl aluminum chloride, and 100 mmoles of titanium trichloride (TAC-131, a product of Toho Titanium Co., Ltd.). The partial hydrogen pressure in the vapor phase was maintained at 2.7 kg/cm², and the polymerization temperature, at 70° C. The polymerization solution was continuously withdrawn from the reaction vessel so that the amount of the solution residing in the vessel became 100 liters. Methanol was added in an amount of 10 liters per hour. The polymer solution

was then washed with water to remove the unreacted monomers. A butene-1/ethylene copolymer was obtained in an amount of 7.3 kg per hour. The results are shown in Table 1.

REFERENTIAL EXAMPLE 9

Referential Example 8 was repeated except that the amount of ethylene charged and the partial hydrogen pressure were changed as indicated in Table 1. The results are shown in Table 1.

while maintaining the temperature of the resin at 240° C. The crystalline random propylene copolymer composition and the crystalline polypropylene were coextruded from the die to form a composite film composed of a substrate layer (40 microns) of the crystalline polypropylene and a layer (10 microns) of the crystalline random propylene copolymer composition. The properties of the composite polypropylene film were evaluated, and the results are shown in Table 3.

TABLE 1

Refer- ential		Amounts of the monomers fed		Ethylene	Intrinsic viscosity	DSC melt- ing point	Crystal-		$[\mathbf{W}_1]$		
		1-Butene	Ethylene	content	$[\eta]$	[Tm]	linity		0.03a + 0.5	[W ₂]	
Example	е	(kg/hr)	(kg/hr)	(mole %)	(dl/g)	(°C.)	(%)	(wt %)	(wt %)	(wt %)	$5 \times [\eta]^{-1.2}$
1	BEC-1	50	0.18	2.5	1.8	115	48.6	0.26	0.58	1.0	2.5
2	BEC-2	**	0.45	7.1	2.3	106	32.4	0.38	0.71	0.7	1.8
3	BEC-3	"	0.70	11.0	2.6	102	25.8	0.41	0.83	0.7	1.6
4	BEC-4	"	1.1	17.2	2.2	97	15.5	0.45	1.02	0.9	1.9
5	BEC-5	"	1.4	22.7	2.4	92	12.8	0.48	1.18	0.9	1.7
6	BEC-1*	"	0.04	0.5	2.4	124	53.6	0.18	0.52	0.5	1.7
7	BEC-2*	**	4.0	60.2	2.0	24	0.3	1.20	2.31	2.0	2.2
8	BEC-3*	"	0.3	4.8	2.1	104	30.6	2.28	0.64	2.6	2.1
9	BEC-4*	**	0.6	10.2	1.9	95	21.3	2.86	0.81	3.1	2.3
		•				Stress	Stress				

Refer- ential Example		Stress Stress at yield at break point point (kg/cm ²) (kg/cm ²)		Elongation at break (%)	Torsional rigidity (kg/cm ²)	Young's modulus (kg/cm ²)	Standard deviation (moles %)
1	BEC-1	140	520	580	820	2000	0.5
2	BEC-2	80	360	610	490	1200	1.5
3	BEC-3	65	240	620	380	950	2.4
4	BEC-4	50	150	680	190	430	3.5
5	BEC-5	40	110	720	90	200	4.6
6	BEC-1*	170	510	380	1300	2000	0.1
7	BEC-2*	3	5	830	5	10	9.2
8	BEC-3*	38	160	. 450	180	410	3.1
9	BEC-4*	17	110	560	90	190	6.5

^{*}Outside the scope of the 1-butene random copolymer of this invention

TABLE 2

	Como	nomer	Intrinsic viscosity	DSC melt-	Crystal-
	Туре	Content (mole %)	[η] (dl/g)	ing point (°C.)	linity (%)
PP-1		0	3.2	163	65
PP-2	_	0	2.3	165	67
PP-3	_	0	3.3	163	64
PEC-1	ethylene	6.4	2.3	134	52
PEB-1	ethylene	3.1	2.5	130	49
	1-butene	2.5			

EXAMPLES 1-8 AND COMPARATIVE EXAMPLES 1-8

Pellets of the crystalline random propylene copolymer PEC-1 or PEB-1 shown in Table 2 and pellets of the 1-butene random copolymer (ii) shown in Table 3 in the proportions indicated were mixed by a Henschel mixer for 1 minute. The mixture was kneaded in the molten state in a melt-extruder to form a melt of a crystalline random propylene copolymer composition. The composition was fed into a die for a two-layer film while maintaining the temperature of the resin at 240° C. Separately, the crystalline polypropylene as a substrate layer shown in Table 3 was melted in another extruder, and fed into the aforesaid two-layer film die

EXAMPLES 9-11 AND COMPARATIVE EXAMPLE 9

Each of the crystalline polypropylene substrates shown in Tables 2 and 3 was melted in an extruder, extruded from a T-die at a resin temperature of 270° C., and cooled and solidified into a sheet form. The extrudate was then passed through hot rolls to stretch it in the machine direction at a stretch ratio of 5 and thus form a monoaxially stretched sheet of crystalline polypropylene.

Each of the crystalline random propylene copolymer compositions shown in Table 3 was melt-kneaded in another extruder, and extruded from another T-die at a resin temperature of 250° C. onto one surface of the resulting monoaxially stretched sheet of crystalline polypropylene. The resulting composite sheet was continuously passed through a heated tenter to stretch it in the transverse direction at a stretch ratio of 10. As a result, a composite polypropylene film was obtained which was composed of a biaxially stretched substrate layer (30 microns) of crystalline polypropylene and a monoaxially stretched layer (5 microns) of the crystalline random propylene copolymer composition. The properties of the resulting composite polypropylene film were evaluated, and the results are shown in Table 3.

TABLE 3

Properties of the composite polypropylene laminate

TABLE 3-continued

Example	copolymer composition				Sub-	Slip coefficient						(8	(g/cm)				
(Ex.) or	Crystalli	ine	Crystalline			strate		Af-	Af-	((coefficient of static friction			riction	/	Af-	
Compara-	randon	n	random Weight		poly-		ter	ter	coefficient of dynamic friction)					ter			
tive Exam-	propyle	пе	1-bute	ne co-	ratio	pro-	Be-	1	7			Ai	fter	Af	fter	1	Afte
ple (CEx.)	copolyme	r (i)	polym	er (ii)	(i)/(ii)	pylene	fore	day	days	Bei	fore	1 0	lay	7 d	lays	day	7 day
Ex. 1	PEC-1	l	BE		50/50	PP-1	2.1	2.3	2.6	0.38	0.36	0.33	0.30	0.37	0.3	5 2.1	2.5
Ex. 2	,,		BEC		,,,	"	1.8	2.1	2.6	0.39	0.35	0.33	0.30	0.38	0.35	5 2.5	3.1
Ex. 3			BE		30/70	"	1.8	2.2	2.7	0.42	0.38		0.35	0.48	0.43		4.4
Ex. 4	"		BEG		50/50	"	1.7	2.0	2.7	0.40	0.37	0.38	0.34	0.42	0.37		3.5
Ex. 5	"		BE		**	#	1.8	2.0	2.7	0.43	0.38	0.39	0.36	0.45	0.39	•	4.8
Ex. 6			BE		"	,,	1.7	2.2	2.8	0.41	0.37	0.38	0.35	0.46	0.39		5.0
Ex. 7	PEB-1	•	BEG		**	"	1.7	2.2	2.7	0.37	0.34		0.32	0.38	0.34	1 2.2	2.9
Ex. 8			"		30/70		1.6	2.5	3.1	0.43	0.39	0.40	0.36	0.46	0.43	1 3.3	4.1
Ex. 9	PEC-1		,,		50/50	PP-3	1.6	2.0	2.5	0.39	0.36	0.35	0.33	0.40	0.36	5 2.1	2.6
Ex. 10			"		30/70	11	1.5	1.8	2.5	0.42	0.38	0.37	0.35	0.44	0.39	3.5	4.5
Ex. 11	PEB-1		"		50/50		1.5	1.9	2.4	0.36	0.34	0.33	0.30	0.36	0.35	5 2.5	3.0
CEx. 1	PEC-1			·	100/00	PP-1	1.2	1.4	1.6	0.33	0.30		0.25	0.30	0.28		0.9
CEx. 2			BEG		0/100	**	1.5	2.9	4.7	0.59	0.49	0.90	0.77	*	_*		12.0
CEx. 3	PEC-1		BEC	-	50/50	**	4.1	4.5	5.0	0.41	0.36	0.38	0.32	0.42	0.38	•	1.1
CEx. 4			BEC		"	**	2.3	3.9	6.3	0.59	0.48	0.88	0.75		 ≉	7.5	12.4
CEx. 5			BEC		"	"	2.4	5.0	7.5	0.47	0.42	0.86	0.75	1	*	7.2	11.9
CEx. 6	···		BEC	•	**	"	2.3	6.3	9.0	0.50	0.42	0.96	0.83	#	— *	8.0	18.9
CEx. 7	PP-2		BEC		"	"	2.0	2.3	2.8	0.33	0.30	0.27	0.25	0.29	0.26	5 2.0	2.6
CEx. 8	,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,,				30/70	"	1.8	2.1	2.6	0.39	0.35	0.34	0.31	0.38	0.34		3.9
CEx. 9	PEC-1		BEC	C-3*	50/50	PP-3	1.7	3.9	5.8	0.45	0.40	0.79	0.70	*	*	5.9	10.1
Example			·····		Pro	perties of	the con	nposite	e poly	propyle	ene lan	niante		- - - - - 	` 	·	
(Ex.) or											Hea	t seal st	rength a	after h	eat-tr	reatment	
Compara-	Canadalla			Heat se	al strengt	h (g/15 n	nm)						(g/15	mm)			
_	Scratch	***************************************			<u></u>	Name of the last o		······································									
tive Exam-	resistance	70°	80°			V		······································		80° 9	90°						
tive Exam- ple (CEx.)	_	70° C.	80° C.	90° C.		110° C.	120° C.	130°				100° C.	110° C	. 120	° C.	130° C.	140°
ple (CEx.) Ex. 1	resistance (%) 2.8	C. 80	C. 200	90° C. 680	100° C. 1130	110° C. 1590	120° C. 1600	165	C. 50	C.	C. 1	700	1620	157	70	130° C. 1690	·
ple (CEx.) Ex. 1 Ex. 2	resistance (%) 2.8 3.1	C. 80 190	C. 200 580	90° C. 680 1100	100° C. 1130 1550	110° C. 1590 1620	120° C. 1600 1600	165 163	C. 50 30	C. 60 2 100 4	C . 1				70		168
ple (CEx.) Ex. 1 Ex. 2 Ex. 3	resistance (%) 2.8 3.1 3.7	C. 80 190 320	C. 200 580 700	90° C. 680 1100 1500	100° C. 1130 1550 1520	110° C. 1590 1620 1590	120° C. 1600 1600 1630	165 163 160	C. 50 30 00	C. 60 2 100 4	C. 1	700	1620	157	70 70	1690	168 165
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4	resistance (%) 2.8 3.1 3.7 3.5	C. 80 190 320 190	200 580 700 530	90° C. 680 1100 1500 1380	100° C. 1130 1550 1520 1550	110° C. 1590 1620 1590 1530	120° C. 1600 1600 1630 1600	165 163 160 165	C. 50 30 50	C. 60 2 100 4 290 6 230 6	C. 1 230 430 590 550	700 920 1430 1160	1620 1530 1560 1550	157 157 165 159	70 70 70 50	1690 1650	168 165 167 165
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5	resistance (%) 2.8 3.1 3.7 3.5 3.3	C. 80 190 320 190 270	200 580 700 530 820	90° C. 680 1100 1500 1380 1550	100° C. 1130 1550 1520 1550 1580	110° C. 1590 1620 1590 1530 1550	120° C. 1600 1600 1630 1630	165 163 165 165	C. 50 30 50 50	C. 60 2 100 4 290 6 350 7	C. 1 230 130 590 550 720	700 920 1430 1160 1100	1620 1530 1560 1550 1590	157 157 165 159 162	70 70 50 90 20	1690 1650 1630	168 165 167 165 165
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9	C. 80 190 320 190 270 510	200 580 700 530 820 1050	90° C. 680 1100 1500 1380 1550 1580	100° C. 1130 1550 1520 1550 1580 1500	110° C. 1590 1620 1590 1530 1550 1620	120° C. 1600 1600 1630 1630 1630	165 163 165 165 160	C. 50 30 50 50 50	C. 60 2 100 4 290 6 350 7 460 8	C. 1 230 430 590 550 720	700 920 1430 1160 1100 1130	1620 1530 1560 1550 1590 1530	157 157 165 162 160	70 70 70 50 90 20	1690 1650 1630 1690 1650 1590	168 165 167 165 165 163
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0	C. 80 190 320 190 270 510 200	200 580 700 530 820 1050 610	90° C. 680 1100 1500 1380 1550 1580 1150	100° C. 1130 1550 1520 1550 1580 1500 1530	110° C. 1590 1620 1590 1530 1550 1620 1590	120° C. 1600 1600 1630 1630 1630 1650	165 163 160 160 160 170	C. 50 30 50 50 50	C. 60 2 100 4 290 6 350 7 460 8 100 5	C. 1 230 430 590 550 720 310	700 920 1430 1160 1100 1130 1030	1620 1530 1560 1550 1590 1530 1580	157 157 165 159 162	70 70 70 50 90 20	1690 1650 1630 1690 1650	1680 1650 1650 1650 1630
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 6 Ex. 7 Ex. 8	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0 3.9	C. 80 190 320 190 270 510 200 280	200 580 700 530 820 1050 610 620	90° C. 680 1100 1500 1380 1550 1580 1150 1530	100° C. 1130 1550 1520 1550 1580 1500	110° C. 1590 1620 1590 1530 1550 1620	120° C. 1600 1600 1630 1630 1630	165 163 165 165 160	C. 50 30 50 50 50	C. 60 2 100 4 290 6 350 7 460 8 100 5	C. 1 230 430 590 550 720	700 920 1430 1160 1100 1130	1620 1530 1560 1550 1590 1530	157 157 165 162 160	70 70 70 50 20 20 30	1690 1650 1630 1690 1650 1590	168 165 167 165 165 163
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 6 Ex. 7 Ex. 8 Ex. 9	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0 3.9 3.0	C. 80 190 320 190 270 510 200 280 270	200 580 700 530 820 1050 610 620 410	90° C. 680 1100 1500 1380 1550 1580 1530 1530 540	100° C. 1130 1550 1520 1550 1580 1500 1530	110° C. 1590 1620 1590 1530 1550 1620 1590	120° C. 1600 1600 1630 1630 1630 1650 1620 580	165 163 160 160 160 170	C. 50 30 50 50 50 50	C. 60 200 290 230 350 460 8 100 300 6	C. 1 230 430 590 550 720 310	700 920 1430 1160 1100 1130 1030	1620 1530 1560 1550 1590 1530 1580	157 157 165 162 162 162 162	70 70 70 50 20 20 30	1690 1650 1630 1690 1650 1590 1630	168 165 167 165 165 165 165 160
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0 3.0 4.2	C. 80 190 320 190 270 510 200 280 270 350	200 580 700 530 820 1050 610 620 410 470	90° C. 680 1100 1500 1380 1550 1580 1530 540 510	100° C. 1130 1550 1520 1550 1580 1500 1530 1580 550 550	110° C. 1590 1620 1590 1550 1620 1630 530 550	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570	165 163 160 160 160 160 165 56 57	C. 50 50 50 50 50 50 70	C. 60 2 100 4 290 6 350 7 460 8 100 5 300 6 230 6 430 5	C. 1 230 430 590 550 720 310 500 530	700 920 1430 1160 1100 1130 1030 1400	1620 1530 1560 1550 1590 1530 1580 1580 550 560	157 157 165 165 166 166 166 166 56	70 70 70 50 20 20 30 20	1690 1650 1630 1690 1650 1590 1630 1670	168 165 167 165 165 163 166 160 56
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 10 Ex. 11	resistance (%) 2.8 3.1 3.7 3.5 3.9 3.0 3.9 3.0 4.2 3.3	C. 80 190 320 190 270 510 200 280 270	200 580 700 530 820 1050 610 620 410	90° C. 680 1100 1500 1380 1550 1580 1530 1530 540	100° C. 1130 1550 1520 1550 1580 1500 1530 1580 550	110° C. 1590 1620 1590 1530 1550 1620 1630 530	120° C. 1600 1600 1630 1630 1630 1650 1620 580	165 163 160 160 160 160 165 56 57	C. 50 50 50 50 50 50 70	C. 60 100 290 230 350 460 300 300 230 430 430	C. 1 230 430 590 550 720 310 500 530 520	700 920 1430 1160 1100 1130 1030 1400 520	1620 1530 1560 1550 1590 1530 1580 1580 550	157 157 165 165 166 166 166 56	70 70 70 50 20 20 30 30	1690 1650 1630 1690 1650 1590 1630 1670 570	168 165 167 165 165 165 160 56 55
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1	resistance (%) 2.8 3.1 3.7 3.5 3.9 3.0 3.9 3.0 4.2 3.3 2.1	80 190 320 190 270 510 200 280 270 350 280	200 580 700 530 820 1050 610 620 410 470	90° C. 680 1100 1500 1380 1550 1580 150 1530 540 510 530	100° C. 1130 1550 1520 1550 1580 1500 1530 1580 550 550	110° C. 1590 1620 1590 1550 1620 1630 530 550	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570	165 163 160 160 160 160 165 56 57	C. 50 50 50 50 50 50 50 50 50 50 50 50 50	C. 60 100 290 230 350 460 300 300 300 430 430	C. 1 230 430 590 550 720 530 530 530	700 920 1430 1160 1100 1130 1030 1400 520 550	1620 1530 1560 1550 1590 1530 1580 1580 550 560	157 157 165 165 166 166 166 56	70 70 70 50 20 20 30 30	1690 1650 1630 1690 1650 1590 1630 1670 570 560	1680 1670 1650 1650 1650 1650 560 550 540
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 11 CEx. 1	resistance (%) 2.8 3.1 3.7 3.5 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9	C. 80 190 320 190 270 510 200 280 270 350	200 580 700 530 820 1050 610 620 410 470	90° C. 680 1100 1500 1380 1550 1580 1530 540 510	100° C. 1130 1550 1520 1580 1580 1580 550 550 560 1380	110° C. 1590 1620 1590 1550 1620 1590 1630 530 550 570 1380	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570 550 80 1350	163 163 163 163 163 163 163 56 57 55 30 137	C. 50 50 50 50 50 50 70 50 70	C. 60 200 290 230 350 460 300 300 230 430 250 5	C. 1 230 430 590 550 720 530 530 530	700 920 1430 1160 1100 1130 1030 1400 520 550	1620 1530 1560 1550 1590 1530 1580 1580 550 560 550	157 157 165 165 166 166 166 56	70 70 70 50 20 20 30 30 40	1690 1650 1630 1690 1650 1590 1630 1670 570 560 540	1680 1650 1650 1650 1650 1600 560 550 540 790
Ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1 CEx. 2 CEx. 2	resistance (%) 2.8 3.1 3.7 3.5 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9 2.5	C. 80 190 320 190 270 510 200 280 270 350 280 —650 —650	200 580 700 530 820 1050 610 620 410 470 420 	90° C. 680 1100 1500 1380 1550 1580 150 540 510 530 1370	100° C. 1130 1550 1520 1580 1580 1580 550 550 560 1380 80	110° C. 1590 1620 1590 1550 1620 1590 1630 530 550 570	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570 550 80	165 165 166 166 166 170 165 56 57 55	C. 50 50 50 50 50 50 70 50 70	C. 60 200 290 230 350 460 300 300 230 430 250 5	C. 1 230 430 590 550 720 530 530 530	700 920 1430 1160 1100 1130 1400 520 550 550	1620 1530 1560 1550 1590 1530 1580 1580 550 560 550	157 157 165 165 166 166 158 166 56 56	70 70 70 50 20 20 30 30 40	1690 1650 1630 1690 1650 1590 1630 1670 570 560 540 120	168 165 167 165 165 165 160 56 55 54 79 138
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1 CEx. 2 CEx. 3 CEx. 3	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9 2.5 6.8	80 190 320 190 270 510 200 280 270 350 280	200 580 700 530 820 1050 610 620 410 470 420	90° C. 680 1100 1500 1380 1550 1580 150 1530 540 510 530	100° C. 1130 1550 1520 1580 1580 1580 550 550 560 1380	110° C. 1590 1620 1590 1550 1620 1590 1630 530 550 570 1380	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570 550 80 1350	163 163 163 163 163 163 163 56 57 55 30 137	C. 50 50 50 50 50 50 70 50 70 70	C. 60 100 290 230 350 460 300 230 430 250 300 6 300 6 300 6 6 6 6 6 6 6 6 6 6 6	C. 1 230 430 590 550 720 530 530 530	700 920 1430 1160 1100 1130 1400 520 550 550	1620 1530 1560 1550 1590 1530 1580 550 560 550	157 157 165 165 166 166 158 167 56 56 54	70 70 70 70 70 70 70 70 70	1690 1650 1630 1690 1650 1590 1630 1670 570 560 540 120 1380	168 165 167 165 165 165 160 56 55 54 79 138 163
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1 CEx. 2 CEx. 3 CEx. 3 CEx. 4 CEx. 5	resistance (%) 2.8 3.1 3.7 3.5 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9 2.5 6.8 4.1	C. 80 190 320 190 270 510 200 280 270 350 280 650 90 120	200 580 700 530 820 1050 610 620 410 470 420 980 220 320	90° C. 680 1100 1500 1380 1550 1580 1530 540 510 530 1370 610 890	100° C. 1130 1550 1520 1580 1580 1580 550 550 560 1380 80	110° C. 1590 1620 1590 1550 1620 1590 1630 530 550 570 1380 380	120° C. 1600 1600 1630 1630 1650 1650 580 570 550 80 1350 920	163 163 163 163 163 163 163 56 57 57 55 30 137 168	C. 50 50 50 50 50 50 50 50 50 50 50 50 50	C. 60 100 290 230 350 460 300 230 430 250 300 6 300 6 300 6 6 6 6 6 6 6 6 6 6 6	C. 1 230 430 590 550 720 530 530 530 530 530 530	700 920 1430 1160 1100 130 1400 520 550 550 550	1620 1530 1560 1550 1590 1580 1580 550 560 550 560 550	157 157 163 163 163 163 163 56 56 54 57 79	70 70 70 70 70 70 70 70 70 70 70 70 70 7	1690 1650 1630 1690 1650 1670 570 560 540 120 1380 1050	168 165 167 165 165 165 160 56 55 54 79 138 163 145
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1 CEx. 1 CEx. 2 CEx. 3 CEx. 3 CEx. 4 CEx. 5 CEx. 5	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9 2.5 6.8	C. 80 190 320 190 270 510 200 280 270 350 280 — 650 — 90	200 580 700 530 820 1050 610 620 410 470 420 980 220	90° C. 680 1100 1500 1380 1550 1580 150 540 510 530 1370 610	100° C. 1130 1550 1520 1580 1580 1580 1580 550 560 560 1380 80 920	110° C. 1590 1620 1590 1550 1620 1590 1630 550 570 1380 380 1290	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570 550 80 1350 920 1430	163 163 163 163 163 163 163 56 57 55 30 137 168 148	C. 50 50 50 50 50 50 50 50 50 50 50 50 50	C. 60 2 100 2 290 6 230 6 350 6	C. 1 230 430 590 550 720 530 530 530 530 530 530 530 530 530	700 920 1430 1160 1100 130 1400 520 550 550 550 550	1620 1530 1560 1550 1590 1580 1580 550 560 550 560 550	157 157 162 162 162 162 162 56 56 54 79 79	70 70 70 70 70 70 70 70 70 70 70 70 70 7	1690 1650 1630 1650 1590 1630 1670 570 560 540 120 1380 1050 1100	168 165 167 165 165 165 160 56 55 54 79 138 163 163 163 163
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1 CEx. 2 CEx. 3 CEx. 3 CEx. 4 CEx. 5 CEx. 5 CEx. 6 CEx. 7	resistance (%) 2.8 3.1 3.7 3.5 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9 2.5 6.8 4.1	C. 80 190 320 190 270 510 200 280 270 350 280 650 90 120	200 580 700 530 820 1050 610 620 410 470 420 980 220 320	90° C. 680 1100 1500 1380 1550 1580 1530 540 510 530 1370 610 890	100° C. 1130 1550 1520 1550 1580 1580 1580 550 560 560 1380 80 920 1550	110° C. 1590 1620 1590 1550 1620 1590 1630 550 570 1380 380 1290 1580	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570 550 80 1350 920 1430 1630	163 163 163 163 163 163 163 163 163 163	C. 50 30 50 50 50 50 50 50 50 50 50 50 50 50 50	C. 60 2 100 2 290 6 230 6 350 6	C. 1 230 430 590 500 530 530 530 530 530 530 530 530 53	700 920 1430 1160 1100 130 1400 520 550 550 550 	1620 1530 1560 1550 1590 1580 1580 550 560 550 560 550 550 550 360	157 157 169 169 169 169 169 56 54 79 79	70 70 70 70 70 70 70 70 70 70 70 70 70 7	1690 1650 1630 1650 1650 1670 570 560 540 120 1380 1050 1100 1050	1680 1650 1650 1650 1650 1650 1650 1650 165
ple (CEx.) Ex. 1 Ex. 2 Ex. 3 Ex. 4 Ex. 5 Ex. 6 Ex. 7 Ex. 8 Ex. 9 Ex. 10 Ex. 11 CEx. 1 CEx. 1 CEx. 2 CEx. 3 CEx. 3 CEx. 4 CEx. 5 CEx. 5	resistance (%) 2.8 3.1 3.7 3.5 3.3 3.9 3.0 3.9 3.0 4.2 3.3 2.1 7.9 2.5 6.8 4.1 5.0	C. 80 190 320 190 270 510 200 280 270 350 280 650 90 120	200 580 700 530 820 1050 610 620 410 470 420 980 220 320	90° C. 680 1100 1500 1380 1550 1580 150 540 510 530 1370 610 890 1570	100° C. 1130 1550 1520 1550 1580 1580 1580 550 560 560 1380 80 920 1550 1550	110° C. 1590 1620 1530 1550 1620 1590 1630 530 550 570 1380 380 1290 1580 1550	120° C. 1600 1600 1630 1630 1630 1650 1620 580 570 550 80 1350 920 1430 1630 1630 1630	163 163 163 163 163 163 163 163 163 163	C. 50 50 50 50 50 50 50 50 50 50 50 50 50	C. 60 2 100 2 290 6 230 6 350 6 300 6	C. 1 230 430 590 500 530 530 530 530 530 530 530 530 53	700 920 1430 1160 1100 130 1400 520 550 550 550 550 220 270	1620 1530 1560 1550 1590 1580 1580 550 560 550 560 550 560 550 360 430	157 157 169 169 169 169 169 56 54 79 79	70 70 70 70 70 70 70 70 70 70 70 70 70 7	1690 1650 1690 1650 1590 1630 1670 570 560 540 120 1380 1050 1100 1050 1120	140° 1680 1650 1650 1650 1650 1650 1630 1630 1630 1630 1630 1650 1650 1280

Note:

-*means that the measurement was impossible.

What is claimed is:

- 1. A crystalline random propylene copolymer composition comprising (i) a crystalline random propylene copolymer comprising 99 to 85 mole % of propylene and 1 to 15 mole % of an alpha-olefin other than propylene and (ii) a random 1-butene copolymer comprising 55 ethylene and 1-butene, the weight ratio of the copolymer (i) to the copolymer (ii) being in the range of from 5:95 to 90:10, said random 1-butene copolymer (ii) satisfying the following conditions:
 - (A) it comprises 1 to 50 mole % of the ethylene com- 60 kg/cm².

 ponent and 50 to 99 mole % of the 1-butene component,

 copolym
 - (B) it has an intrinsic viscosity $[\eta]$, measured in decalin at 135° C., of from 0.5 to 6 dl/g,
 - (C) it has a melting point [Tm], measured by a differ- 65 ential scanning calorimeter, of from 30° to 130° C.,
 - (D) it has a crystallinity, measured by X-ray diffractometry, of from 1 to 60%,

- (E) the amount [W₁ in % by weight] of that portion of it which is soluble in boiling methyl acetate is not more than 2%, and
- (F) the amount [W₂ in % by weight] of that portion of it which is soluble in a 1:1 by volume mixture of acetone and n-decane at 10° C. is less than $5 \times [\eta]^{-1.2}$.
- 2. The composition of claim 1 wherein the 1-butene copolymer further has a stress at yield point (G), measured in accordance with JIS K-7113, of 10 to 200 kg/cm².
- 3. The composition of claim 1 wherein the 1-butene copolymer further has a stress at break point (H), measured in accordance with JIS K-7113, of 3 to 1000 kg/cm².
- 4. The composition of claim 1 wherein the 1-butene copolymer further has an elongation at break point (I), measured in accordance with JIS K-7113, of at least 300%.

- 5. The composition of claim 1 wherein the 1-butene copolymer further has a torsional rigidity (J), measured in accordance with JIS K-6745, of 5 to 3000 kg/cm².
- 6. The composition of claim 1 wherein the 1-butene copolymer has a Young's modulus (K), measured in 5 accordance with JIS K-7113, of 10 to 5000 kg/cm².
- 7. The composition of claim 1 wherein the standard deviation (L) of the 1-butene content of the 1-butene copolymer is not more than 0.4a mole in which a represents the ethylene content of said copolymer.
- 8. The composition of claim 1 wherein the 1-butene copolymer is composed of 60 to 99 mole % of the 1butene component and 1 to 40 mole % of the ethylene.
- 9. The composition of claim 1 wherein the weight ratio of the crystalline random propylene copolymer (i) 15 to the 1-butene-type random copolymer (ii) is in the range of from 10:90 to 85:15.
- 10. A polypropylene composite laminate comprising a substrate layer of crystalline propylene and a layer of the crystalline random propylene copolymer composi- 20 tion of claim 1 laminated to at least one surface of said substrate layer.
- 11. The laminate of claim 10 wherein the substrate layer of crystalline polypropylene is in the nonstretched state.
- 12. The laminate of claim 10 wherein the layer of the crystalline random propylene copolymer is in the nonstretched state.
- 13. The laminate of claim 10 wherein the layer of the crystalline random propylene copolymer composition 30 has a thickness of 0.1 to 50 microns.
- 14. The laminate of claim 10 wherein the crystalline polypropylene constituting the substrate layer is selected from crystalline homopolymer of propylene and crystalline propylene/alpha-olefin random copolymers. 35
- 15. The composition of claim 1 wherein the amount of that portion of the 1-butene-type copolymer (ii) which is soluble in a mixture of acetone and n-decane at 10° C. is $0.1 \times [\eta]^{-1.2}$ to $5 \times [\eta]^{-1.2}\%$ by weight.
- 16. The composition of claim 1 wherein the amount 40 or biaxially stretched state. of that portion of the 1-butene-type copolymer (ii) which is soluble in a mixture of acetone and n-decane at 10° C. is $0.2 \times [\eta]^{-1.2}$ to $4.5 \times [\eta]^{-1.2}\%$ by weight.
- 17. The composition of claim 1 wherein the amount of that portion of the 1-butene-type copolymer (ii) 45 which is soluble in a mixture of acetone and n-decane at 10° C. is $0.3 \times [\eta]^{-1.2}$ to $4 \times [\eta]\%$ by weight.

- 18. The laminate of claim 10 wherein the amount of that portion of the 1-butene-type copolymer (ii) which is soluble in a mixture of acetone and n-decane at 10° C. is $0.1 \times [\eta]^{-1.2}$ to $5 \times [\eta]^{-1.2}\%$ by weight.
- 19. The laminate of claim 10 wherein the amount of that portion of the 1-butene-type copolymer (ii) which is soluble in a mixture of acetone and n-decane at 10° C. is $0.2 \times [\eta]^{-1.2}$ to $4.5 \times [\eta]^{-1.2}\%$ by weight.
- 20. The laminate of claim 10 wherein the amount of 10 that portion of the 1-butene-type copolymer (ii) which is soluble in a mixture of acetone and n-decane at 10° C. is $0.3 \times [\eta]^{-1.2}$ to $4 \times [\eta]^{-1.2}\%$ by weight.
 - 21. The composition of claim 1 wherein the crystalline random propylene copolymer (i) comprises 98 to 90 mole percent of propylene and 2 to 10 mole percent of the alpha-olefin other than propylene.
 - 22. The composition of claim 21 wherein the alphaolefin other than propylene is ethylene, 1-butene, 1-pentene, 1-hexene, 4-methyl-1-pentene or a mixture of at least two of these alpha-olefins.
 - 23. The composition of claim 1 wherein the random 1-butene copolymer (ii) satisfies the following conditions:
 - (A) it comprises 1 to 40 mole percent of ethylene and 60 to 99 mole percent of 1-butene,
 - (B) it has an intrinsic viscosity (η) , measured in decalin at 135° C., of 1.7 to 3.5 dl/g,
 - (C) it has a melting point (Tm), measured by a differential scanning calorimeter, of from 120° to 145°
 - (D) it has a crystallinity, measured by X-ray diffractometry, of from 1 to 55%,
 - (E) the amount (W_1 in percent by weight) of that portion of it which is soluble in boiling methylacetate is from 0.02 to 1% by weight, and
 - (F) the amount (W₂ in percent by weight) is from $0.2\times(\eta)^{-1.2}$ to $4.5\times(\eta)^{-1.2}$.
 - 24. The laminate of claim 10 wherein the the substrate layer of crystalline polypropylene is in the monoaxially
 - 25. The laminate of claim 10 wherein the layer of the crystalline random propylene copolymer is in the monoaxially or biaxially stretched state.
 - 26. The laminate of claim 10 wherein a layer of said crystalline random propylene copolymer is laminated to both surfaces of said substrate layer.

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UNITED STATES PATENT AND TRADEMARK OFFICE CERTIFICATE OF CORRECTION

PATENT NO.: 4,734,328

DATED : March 29, 1988

INVENTOR(S): MASAKI KOHYAMA ET AL.

It is certified that error appears in the above-identified patent and that said Letters Patent is hereby corrected as shown below:

In Claim 17, Column 19, line 4 of the Claim, "4X[η]%" should read --4X[η] -1.2%--.

> Signed and Sealed this Second Day of August, 1988

Attest:

DONALD J. QUIGG

Attesting Officer

Commissioner of Patents and Trademarks