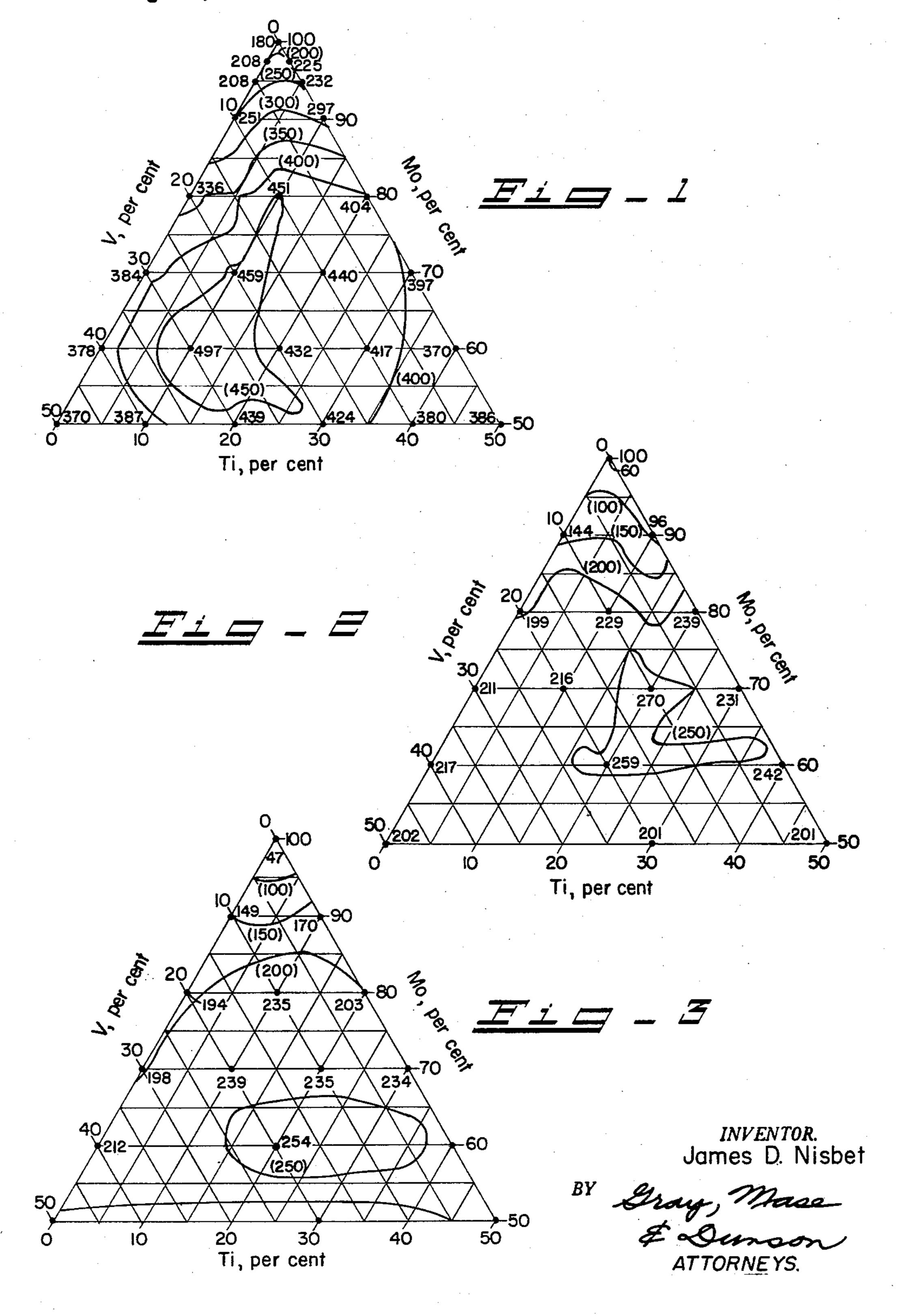
MOLYBDENUM-BASE ALLOY

Filed Aug. 29, 1955

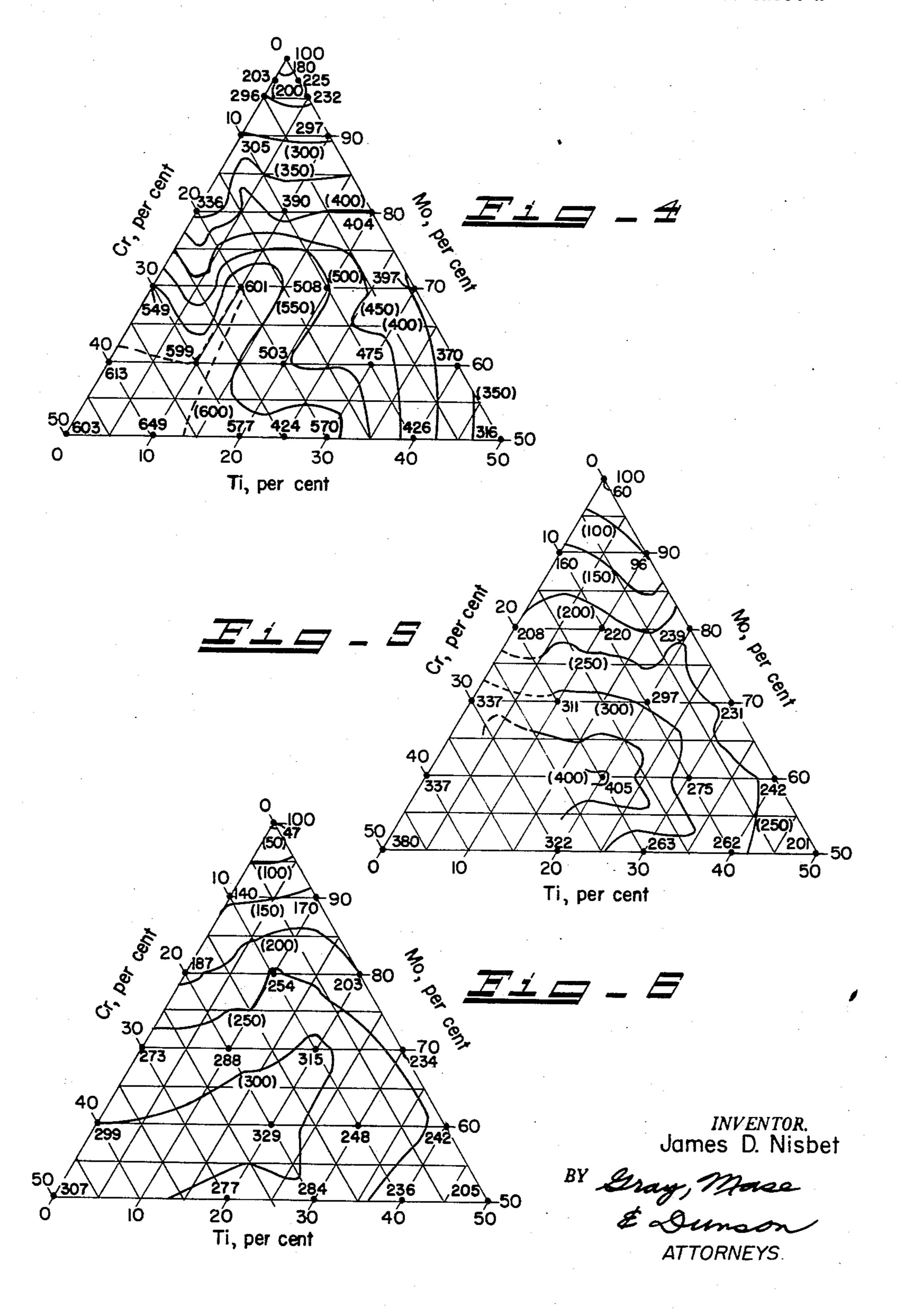
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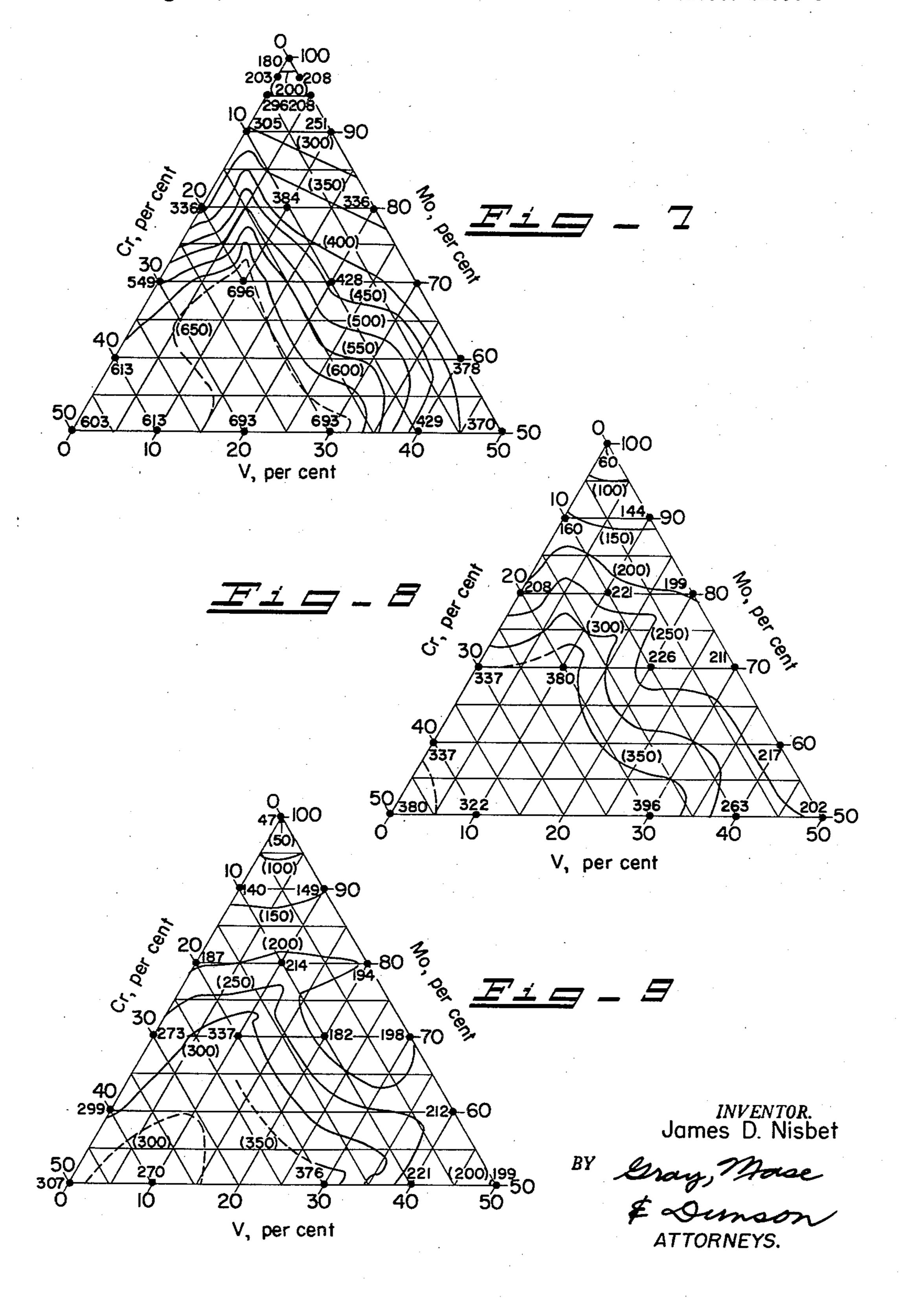
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MOLYBDENUM-BASE ALLOY

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MOLYBDENUM-BASE ALLOY

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Application August 29, 1955, Serial No. 531,056 6 Claims. (Cl. 75—176)

This invention relates to molybdenum-base alloys, and more specifically to molybdenum-base alloys which exhibit superior strength and hardness over a wide range of temperatures.

For a great variety of industrial and military applications, it is highly desirable to use metals that maintain their hardness and strength at elevated temperatures. For such applications as gas turbine components, jet engine parts, high-speed cutting tools and dies there is an ever-increasing demand for hard and strong metals and materials that will retain such hardness and strength at higher temperatures.

Although the alloys used today, such as the cobaltchromium-nickel alloys for high-temperature, highstrength applications are far superior to metals once used for high-temperature applications, the increasing demands for harder and stronger metals for high-temperature applications make any advance in the art toward this goal highly significant.

The principal object of this invention is to provide improved cast alloys which exhibit high strength and hardness at both room and elevated temperatures.

Another object of this invention is to provide molybdenum-base alloys which retain high hardness at elevated temperatures.

Another object of this invention is to provide molybdenum-base alloys which will retain room-temperature hardness properties after being subjected to high temperatures.

Other objects and advantageous features will be obvious in the following specification and examples.

In accordance with the present invention binary, ternary and complex molybdenum-base alloys have been found that possess improved strength properties at elevated temperatures.

In general, this invention relates to molybdenum-base alloys containing at least one metal selected from the high-melting-point refractory metals from the IV-A, V-A, and VI-A atomic groups of the periodic table. Specifically, this invention relates to hard, high-melting-point alloys which possess high strength or hardness at room and elevated temperatures and which contain at least 50 percent molybdenum and at least one metal selected from the group tantalum, titanium, columbium, vanadium, tungsten, and chromium.

In the drawings:

Fig. 1 is a triaxial diagram on which is plotted compositions and hardness levels of binary and ternary molybdenum-base alloys containing vanadium and/or titanium. The hardnesses are room-temperature Vickers 65 hardness numbers.

Fig. 2 is a triaxial diagram on which are plotted compositions of molybdenum-base alloys containing vanadium and/or titanium as in Fig. 1 but the Vickers hardness numbers of Fig. 2 are taken at 700° C.

Fig. 3 is a triaxial diagram on which are plotted compositions of molybdenum-base alloys containing vanadium

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and/or titanium as in Fig. 1 but the Vickers hardness numbers of Fig. 3 are taken at 900° C.

Fig. 4 is a triaxial diagram on which is plotted compositions and hardness levels of binary and ternary molybdenum-base alloys containing chromium and/or titanium. The hardnesses are room temperature Vickers hardness numbers.

Fig. 5 is a triaxial diagram on which are plotted compositions of molybdenum-base alloys containing chromium and/or titanium as in Fig. 4 but the Vickers hardness numbers of Fig. 5 are taken at 700° C.

Fig. 6 is a triaxial diagram on which are plotted compositions of molybdenum-base alloys containing chromium and/or titanium as in Fig. 4 but the Vickers hardness numbers of Fig. 6 are taken at 900° C.

Fig. 7 is a triaxial diagram on which is plotted compositions and hardness levels of binary and ternary molybdenum-base alloys containing chromium and/or vanadium. The hardnesses are room-temperature Vickers hardness numbers.

Fig. 8 is a triaxial diagram on which are plotted compositions of molybdenum-base alloys containing chromium and/or vanadium as in Fig. 7 but the Vickers hardness numbers of Fig. 8 are taken at 700° C.

Fig. 9 is a triaxial diagram on which are plotted compositions of molybdenum-base alloys containing chromium and/or vanadium as in Fig. 7 but the Vickers hardness numbers of Fig. 9 are taken at 900° C.

In the drawings, the top or uppermost point of the triaxial diagram represents 100 percent molybdenum, while the base of the diagram represents 50 percent molybdenum. Alloying metals are designated for each side of the diagrams. The composition of any point on the diagram may be determined by reading molybdenum content from the bottom to the top at 10 percent intervals, as indicated. Alloying metal contents are determined by projecting a line from the point to the side of the diagram representing such addition, parallel to the lines projecting beyond the diagram and read at 10 percent intervals, as indicated. For example, in Fig. 1 titanium content is determined by projecting a line to the bottom of the diagram, as indicated, parallel to the lines extending slightly below the diagram at the bottom and read from left to right. Vanadium content is determined in the same manner as the titanium content but read along the left side of the diagram, as indicated. The points which are marked and Vickers hardness numbers given are actual examples produced in the laboratory and charted on the diagrams. The solid lines are contour lines designating levels of hardness and the circled numbers indicate the hardness levels by Vickers hardness numbers.

Molybdenum-base binary alloys, containing as the alloying addition, titanium, vanadium, chromium, columbium or tantalum have been found to have excellent high-strength or hardness properties at both room temperature and at elevated temperatures. In a like manner, ternary alloys containing at least 50 percent molybdenum and at least two metals selected from the group titanium, vanadium, chromium, tungsten, columbium, and tantalum have been found to have equal or superior strength to the binary system.

Alloys which are suitable for applications such as high-speed cutting tools, dies, jet engine parts, and gas-turbine components preferably have a room-temperature hardness of at least 300 Vickers hardness number and for high-temperature applications a hardness of at least 150 Vickers hardness number. Table I below illustrates the minimum binary alloying additions to molybdenum which will result in minimum room-temperature hardnesses of at last 300 Vickers hardness number and at least 150 Vickers hardness number at a temperature of 700° C.

It is further illustrated by Figs. 1 through 9 and Table II that ternary alloys containing at least 50 percent molybdenum and at least one of any of the above binary additions to molybdenum will present high strength properties.

The properties of three of the binary and three of the ternary alloys are clearly illustrated in Figs. 1 through 9. The range of molybdenum-titanium-vanadium alloys possessing room-temperature hardnesses in excess of 300 Vickers hardness number and 150 Vickers hardness number at 700° C. are clearly illustrated in Table II and Figs. 1 and 2. These ternary alloy compositions are shown in 20 Fig. 2 by all of the compositions falling below the 150 Vickers hardness number level. The range of molybdenum - titanium - chromium alloys possessing roomtemperature hardnesses in excess of 300 Vickers hardness number and 150 Vickers hardness number at 700° C. are 25 clearly illustrated in Figs. 4 and 5. These ternary alloy compositions are shown in Fig. 5 by all the compositions falling below the 150 Vickers hardness number level. The range of molybdenum-vanadium-chromium alloys possessing room-temperature hardnesses in excess of 300 Vickers hardness number and 150 Vickers hardness number at 700° C. are clearly illustrated in Figs. 7 and 8. These ternary alloy compositions are shown in Fig. 8 by all the compositions falling below the 150 Vickers hardness number level line.

In a similar manner, ternary alloys including at least two of the binary additions described above will provide room-temperature hardnesses in excess of 300 Vickers hardness number and high-temperature properties of at least 150 Vickers hardness number at 700° C.

Alloys containing more than 50 percent alloying additions to molybdenum are outside the range of the alloys disclosed and are no longer molybdenum-base alloys. Strength properties generally fall off as the alloying content increases beyond 50 percent. The molybdenum content is, therefore, preferably a minimum of 50 percent by 45 weight of the alloy.

Tungsten may be alloyed to molybdenum as a ternary or complex alloying addition falling within the scope of the present invention, so long as the strength properties at room and elevated temperatures are not adversely af- 50 fected. For examples, as may be noted by referring to Table II, binary molybdenum-tugsten alloys possess hardnesses far below 150 Vickers hardness number at a temperature of 700° C., therefore, binary molybdenumtungsten alloys do not fall in the scope of the present in- 55 vention. Also, as exemplified by Table II, ternary molybdenum-tungsten alloys may possess sufficient hardness properties to fall within the scope of the present invention. However, the alloys of the present invention contain at least 50 percent molybdenum. As can be seen 6 in Table II, large additions of tungsten may be made while retaining high hardness.

It can be seen from Figs. 3, 6, and 9 that there are also optimum ranges for certain molybdenum-base ternary alloys. These alloys exhibit a minimum of hardness of 6 250 Vickers hardness number at 900° C. Though the alloys of this invention retain, to a remarkable extent, their hardness at elevated temperatures, there is, of course, some hardness drop-off as the temperature rises. Although hardnesses of 150 Vickers at temperatures as high 7 as 700° C. are useful high-temperature properties, it is felt that hardnesses of 250 Vickers at temperatures of 900° C. are of much greater significance. Such high-temperature hardness and strength render these alloys, within the optimum composition ranges, ideal materials 7

for high-temperature applications where high-strength properties are required. Compositions exhibiting 250 Vickers or greater hardness at 900° C. are set forth in Figs. 3, 6, and 9 of the accompanying diagrams within the 250 Vickers hardness number levels.

Microscopic studies show that molybdenum-base alloys containing titanium, vanadium, tantalum, columbium, chromium and tungsten are all single phase solid solution alloys. The alloys discussed so far are, therefore, the solid solution type of alloys which exhibit solid solution strength properties.

It has been found that improved room and elevated temperature strength properties may be attained by alloying zirconium with the other molybdenum-base alloys of the present invention. The addition of zirconium to molybdenum or molybdenum-base alloys results in microscopic structures composed of a terminal solid-solution phase of zirconium in molybdenum and the intermetallic compound ZrMo₂. The difficulty experienced with this type of alloy is the complete lack of workability of the metal at either room or elevated temperatures, however, it has been found that small additions of zirconium to the above binary or complex alloys may be satisfactorily employed to attain improved strength properties while retaining the required degree of workability necessary to be adapted to high-temperature applications. The small additions of zirconium to the terminal solid-solution phase molybdenum-base alloys which may be employed to attain the room and high temperature strength properties of the present invention while maintaining adequate workability is from about 2 to 10 percent.

Examples of binary molybdenum-zirconium alloys given in Table II illustrate the extreme hardness of the binary alloy.

It is obvious that combinations including any of the alloying additions to molybdenum of this invention can be made to form alloys containing three and even four alloying elements, without departing from the scope of this invention.

The molybdenum-base alloys of the present invention were arc-melted using a water-cooled hearth. A very pure helium atmosphere at low pressures (10 to 20 cm. of mercury pressure) was employed. By using a pure helium atmosphere, little or no oxygen pickup was experienced so that it was not necessary to add deoxidants such as carbon, aluminum, or boron to the melts. It should be noted that additions of such elements as carbon, aluminum, or boron generally harden the metal to some extent.

Table II below covers all the ranges disclosed above, illustrates specific examples of the molybdenum-base alloys of this invention and shows their strength properties at various elevated temperatures.

Table II

UU	·			·	· · · · · · · · · · · · · · · · · · ·	, ** · · .	
	Composition, weight	Vickers hardness number					
	percent (balance Mo)	25° C.	300° C.	500° O.	700° C.	900° C.	
60	Mo (unalloyed)	180 225	80	74	60	47	
. •	2.5 Ti	232 279	211	155	96	170	
	20 Ti	404 397	292 288	242 250	239 231	203 234	
65	40 Ti	370 316 208	303 236	254 225	242 201	242 205	
	5 V	208 251 336	178 248	151 197	144 199	149 194	
	30 V	384 378	257 268	228 239	211 217	198 212	
70	50 V 2.5 Cr 5 Cr	370 203 296	222	228	202	199	
	10 Cr 20 Cr	305 336	202 260	165 231	160 208	140 187	
	30 Cr 40 Cr 50 Cr	549 613 603	462 506 418	375 412 389	337 337 380	273 299 307	
75	2.5 Cb	201					

Table II.—Continued

Composition, weight	Vickers hardness number					
percent (balance Mo)	25° C.	300° C.	500° C.	700° C.	900° C.	
5 Cb	220					
10 Cb 20 Cb 30 Cb	269 324 .397	173 218 265	154 198 233	132 174 211	134 167	
40 Cb	431 476 207	275 284	267 286	250 278	250 267	I
5 Ta 10 Ta	206 231	137	107	96	108	
20 Ta 30 Ta 40 Ta	302 343 396	191 259 275	181 220 244	150 199 242	168 220 234	•
50 Ta 10 Ti-10 V 20 Ti-10 V	459 451 440	286 284 315	253 256 282	223 229 270	234 235 235	1
10 Ti-20 V 30 Ti-10 V	459 417	250	233	216	239	
10 Ti-30 V 40 Ti-10 V 10 Ti-40 V	497 380 387					2
20 Ti-20 V 30 Ti-20 V 20 Ti-30 V	432 424 439	322 234	299 218	259 201	254 185	
10 Ti-10 Cr	390 508 601	270 357 379	235 352 329	220 297	254 315	
30 Ti-10 Cr 10 Ti-30 Cr	475 599	286	277	311 275	288 248	2
40 Ti-10 Cr	426 649 503	297 383	259 379	262 405	236 329	
20 Ti-30 Cr 30 Ti-20 Cr 10 Cr-10 V	570 577 384	379 370 247	379 383 248	322 263 221	277 284 214	
20 Cr-10 V 10 Cr-20 V	696 428	437 247	398 190	380 226	337 182	3
40 Cr-10 V 10 Cr-40 V 30 Cr-20 V	613 429 693	288 290	250 271	322 263	270 221	
20 Cr-30 V 25 Cr-25 Cb 25 Cr-25 Ta	693 645 371	516	456	396	376	•
25 Cr-25 Ti 25 Cb-25 Ta 25 Cb-25 Ti	424 435 414					3
25 Cb-25 W	393 425					
25 Ta-25 V 25 Ta-25 W 25 Ti-25 W	473 361 427					4
25 V-25 W	372 213 208	93 106	71 97	71 90	63 84	_
30 W	$\begin{array}{c} 213 \\ 242 \end{array}$	106 111	76 108	71 94	71 89	
50 W 1 Zr 2.5 Zr	244 208 314	159	127	106	77	4
5 Zr	384 363 519					
	0.10					

What is claimed is:

1. An alloy consisting essentially of at least 50 percent molybdenum, and two metals selected from the group consisting of 10 to 40 percent tungsten, 10 to 40 percent titanium, 10 to 40 percent vanadium, 10 to 40 percent chromium, 10 to 40 percent columbium, and 10 to 40 percent tantalum, and characterized by high strength properties.

2. Alloys consisting essentially of at least 50 percent molybdenum, from 10 to 40 percent titanium, from 10 to 40 percent vanadium, and characterized by higher strength properties than binary molybdenum-titanium and molyb-

denum-vanadium alloys.

3. Alloys consisting essentially of at least 50 percent molybdenum, from 10 to 40 percent titanium, and from 10 to 40 percent chromium.

4. Alloys consisting essentially of at least 50 percent molybdenum, from 10 to 40 percent chromium, and from 10 to 40 percent vanadium.

5. Alloys consisting essentially of at least 50 percent molybdenum, from 10 to 40 percent chromium, and from 10 to 40 percent tantalum.

6. Alloys consisting essentially of at least 50 percent molybdenum, from 10 to 40 percent chromium, and from 10 to 40 percent tungsten.

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